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## TiO<sub>2</sub>-based heterojunction photocatalysts for photocatalytic reduction of CO<sub>2</sub> into solar fuels

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In the twenty-first century, global warming and energy shortage have become major global issues. Up to now, the utilization of CO<sub>2</sub> as a carbon source for the production of fuels and chemicals has received increased attention. The photocatalytic reduction of CO<sub>2</sub> into solar fuels has turned out to become one of the most promising and environmentally friendly methods. Well-defined heterojunction structures between two semiconductors with matching electronic band structures can effectively facilitate charge transfer and suppress the recombination of photogenerated electrons and holes, resulting in enhanced photocatalytic performance. This review focuses on the design and fabrication of TiO<sub>2</sub>-based heterojunction photocatalysts and their recent progresses into developing solar fuels via the photocatalytic reduction of CO<sub>2</sub>. The photocatalytic performances of a number of typical TiO<sub>2</sub>-based heterojunction photocatalysts, e.g., p-n, non-p-n, Z-scheme, TiO<sub>2</sub>-metal, TiO<sub>2</sub>-carbon, phase, facet, and other heterojunctions, are summarized and analyzed. The reaction mode and some typical photoreactors, e.g., slurry photoreactor, optical-fiber photoreactor, monolith photoreactor, and optofluidic microreactor, are also presented and analyzed. In the end, we propose a perspective on the opportunities and challenges to design new types of photocatalysts and photoreactors for improving the photocatalytic reduction of CO<sub>2</sub>.

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# 1. Introduction

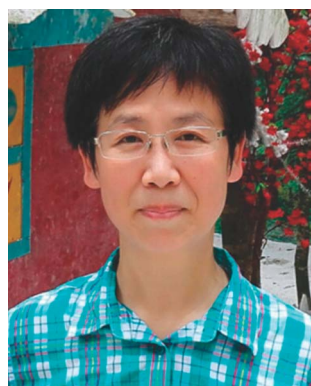
The recent fast-paced developments in the social economy and industrialization have given rise to the problems of environmental pollution and energy shortage. The emission of carbon dioxide (CO<sub>2</sub>) is mainly caused by the combustion of fossil fuels, resulting in a continuous rise in the concentration of CO<sub>2</sub> in the atmosphere.<sup>1–3</sup> As a major greenhouse gas, CO<sub>2</sub> can cause global warming, leading to disastrous climatic conditions.<sup>4</sup> Hence, decreasing the emission of CO<sub>2</sub> and rationally utilizing CO<sub>2</sub> have become formidable and important research subjects around the world. It is well known that CO<sub>2</sub> is a very stable molecule with a nonpolar covalent bond.<sup>5–11</sup> The dissociation energy of C=O bond in the CO<sub>2</sub> molecule is ~750 kJ mol<sup>-1</sup>, which is higher than C–H (~430 kJ mol<sup>-1</sup>) and C–C (~336 kJ mol<sup>-1</sup>) bonds.<sup>12</sup> CO<sub>2</sub> has high thermodynamic stability and kinetic inertness; further, it has very low solubility in water (~30 mM under room temperature and 1 atm).<sup>13–15</sup> Therefore, the activation and conversion of CO<sub>2</sub> pose a big challenge.

In the natural environment, green plants absorb solar energy and use CO<sub>2</sub> and H<sub>2</sub>O as raw materials to manufacture

carbohydrates with the simultaneous release of oxygen.<sup>16–20</sup> Inspired by this natural process, artificial photosynthesis has become a research hotspot, showing rapid developments. In the early stages of this research paradigm, Fujishima and Honda reported certain semiconductor photocatalysts dispersed in water for the photocatalytic reduction of CO<sub>2</sub>.<sup>21</sup> In the past decade, many photocatalysts, *e.g.*, TiO<sub>2</sub>,<sup>22–26</sup> g-C<sub>3</sub>N<sub>4</sub>,<sup>27–31</sup> ZnIn<sub>2</sub>S<sub>4</sub>,<sup>32–35</sup> Bi<sub>2</sub>WO<sub>6</sub>,<sup>36–41</sup> graphene (GR),<sup>42–46</sup> CdS,<sup>47–50</sup> SrNb<sub>2</sub>O<sub>6</sub>,<sup>51</sup> and ZnO,<sup>52–54</sup> have been investigated for the photocatalytic reduction of CO<sub>2</sub>. Among these photocatalysts, TiO<sub>2</sub> is the most widely used due to its abundance, chemical stability, low cost, and resistance toward corrosion.<sup>55,56</sup> With a wide bandgap of ~3.2 eV, TiO<sub>2</sub> can only absorb UV light, which represents 3–5% of the entire sunlight spectrum; therefore, it significantly limits its utilization of solar light.<sup>57</sup> Many strategies have been used for improving its activity and extending its light absorption in the visible light region.<sup>58–60</sup> Wang *et al.* firstly reported that adding MgO or polyaniline onto Pt–TiO<sub>2</sub> could improve the production of CH<sub>4</sub> for the photocatalytic reduction of CO<sub>2</sub>.<sup>61–63</sup> In another study, Wang *et al.* also reported the core–shell-structured Pt@Cu<sub>2</sub>O/TiO<sub>2</sub> to produce CH<sub>4</sub> and CO for CO<sub>2</sub> reduction.<sup>64</sup>

The well-defined heterojunction between two semiconductors with matching electronic band structures can significantly enhance the separation of photogenerated electrons (e<sup>-</sup>) and holes (h<sup>+</sup>), thereby considerably enhancing the photocatalytic performance.<sup>65–70</sup> Yu *et al.* first synthesized a novel Ag<sub>2</sub>O/Ag<sub>2</sub>CO<sub>3</sub>-heterostructured photocatalyst *via* a one-step phase transformation method. The photocatalytic activity for decomposing methyl orange was 68 and 30 times higher than that of pure Ag<sub>2</sub>CO<sub>3</sub> and Ag<sub>2</sub>O, respectively.<sup>71</sup> Recent studies have found that the unique role of a heterojunction can effectively promote the activity during the photocatalytic reduction of CO<sub>2</sub> into solar fuels.<sup>72–77</sup>

In this review, we will focus on the fundamental principles involved in the photocatalytic reduction of CO<sub>2</sub> and highlight the recent advances in TiO<sub>2</sub>-based heterojunction



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photocatalysts for the photocatalytic reduction of CO<sub>2</sub>. Moreover, the reaction mode and some typical photoreactors, *e.g.*, slurry photoreactor, optical-fiber photoreactor, monolith photoreactor, and optofluidic microreactor, are further presented and analyzed. Future prospects on the opportunities, research directions, and remaining challenges for the photocatalytic reduction of CO<sub>2</sub> will be addressed in the last section. We believe this review can provide some useful guidelines for the further development of highly active heterojunction photocatalysts for the photocatalytic reduction of CO<sub>2</sub>.

## 2. Fundamental understanding for photocatalytic reduction of CO<sub>2</sub>

### 2.1 Thermodynamic analysis

Due to its very stable and low energy grade, the chemical transformation of CO<sub>2</sub> is highly unfavorable in thermodynamics.<sup>78–80</sup> The possible reactions that can occur in the photocatalytic reduction of CO<sub>2</sub> in an aqueous medium are listed in Table 1.<sup>81,82</sup> The enthalpy change ( $\Delta H$ ) of all the reactions are highly positive, indicating that the reactions are endothermic and it becomes difficult for the corresponding reactions to occur at ambient temperatures. The Gibbs free energy change ( $\Delta G$ ) is also highly positive and a spontaneous reaction cannot occur. Moreover, it is evident that the photocatalytic reduction of CO<sub>2</sub> can store more energy than splitting water. Therefore, the photocatalytic reduction of CO<sub>2</sub> into hydrocarbon fuels requires a powerful catalyst and a large energy input, which is a big challenge in CO<sub>2</sub> reduction.

Table 2 lists the thermodynamic potentials of different products for CO<sub>2</sub> reduction with the thermodynamic potentials *vs.* normal hydrogen electrode (NHE) at pH = 7 in an aqueous solution at 25 °C and 1 atm.<sup>83–86</sup> It clearly shows that the thermodynamic potential for the reduction of CO<sub>2</sub> to CO<sub>2</sub><sup>•-</sup> radicals is –1.9 V by one electron, indicating that the reaction is highly unfavorable. Owing to the structural differences between linear CO<sub>2</sub> and bent CO<sub>2</sub><sup>•-</sup> radicals, a large kinetic “overvoltage” for one electron reduction is required (in eqn (1)).<sup>87</sup> It is evident that the other reactions with multiple proton-coupled electron transfer steps have lower standard potentials than one electron

**Table 1** Possible reactions for the photocatalytic reduction of CO<sub>2</sub><sup>81,82</sup>

| Equation | Reaction                                                                                     | $\Delta H^\ominus$<br>(kJ mol <sup>-1</sup> ) | $\Delta G^\ominus$<br>(kJ mol <sup>-1</sup> ) |
|----------|----------------------------------------------------------------------------------------------|-----------------------------------------------|-----------------------------------------------|
| 1        | H <sub>2</sub> O (l) → H <sub>2</sub> (g) + 1/2O <sub>2</sub> (g)                            | 286                                           | 237                                           |
| 2        | CO <sub>2</sub> (g) → CO (g) + 1/2O <sub>2</sub> (g)                                         | 283                                           | 257                                           |
| 3        | CO <sub>2</sub> (g) + H <sub>2</sub> O (l) → HCOOH (l) + 1/2O <sub>2</sub> (g)               | 270                                           | 286                                           |
| 4        | CO <sub>2</sub> (g) + H <sub>2</sub> O (l) → HCHO (l) + O <sub>2</sub> (g)                   | 563                                           | 522                                           |
| 5        | CO <sub>2</sub> (g) + 2H <sub>2</sub> O (l) → CH <sub>3</sub> OH (l) + 3/2O <sub>2</sub> (g) | 727                                           | 703                                           |
| 6        | CO <sub>2</sub> (g) + 2H <sub>2</sub> O (l) → CH <sub>4</sub> (g) + 2O <sub>2</sub> (g)      | 890                                           | 818                                           |

**Table 2** Thermodynamic potentials of CO<sub>2</sub> reduction into various products<sup>83–86</sup>

| Equation | Reaction                                                                                                     | $E^0$ (V)<br><i>vs.</i> NHE at pH = 7 |
|----------|--------------------------------------------------------------------------------------------------------------|---------------------------------------|
| 1        | CO <sub>2</sub> + e <sup>-</sup> → CO <sub>2</sub> <sup>•-</sup>                                             | -1.9                                  |
| 2        | CO <sub>2</sub> + 2H <sup>+</sup> + 2e <sup>-</sup> → CO + H <sub>2</sub> O                                  | -0.53                                 |
| 3        | CO <sub>2</sub> + 4H <sup>+</sup> + 4e <sup>-</sup> → C + 2H <sub>2</sub> O                                  | -0.20                                 |
| 4        | CO <sub>2</sub> + 2H <sup>+</sup> + 2e <sup>-</sup> → HCOOH                                                  | -0.61                                 |
| 5        | CO <sub>2</sub> + 4H <sup>+</sup> + 4e <sup>-</sup> → HCHO + H <sub>2</sub> O                                | -0.48                                 |
| 6        | CO <sub>2</sub> + 6H <sup>+</sup> + 6e <sup>-</sup> → CH <sub>3</sub> OH + H <sub>2</sub> O                  | -0.38                                 |
| 7        | CO <sub>2</sub> + 8H <sup>+</sup> + 8e <sup>-</sup> → CH <sub>4</sub> + 2H <sub>2</sub> O                    | -0.24                                 |
| 8        | 2CO <sub>2</sub> + 9H <sup>+</sup> + 12e <sup>-</sup> → C <sub>2</sub> H <sub>5</sub> OH + 3H <sub>2</sub> O | -0.33                                 |
| 9        | 2H <sup>+</sup> + 2e <sup>-</sup> → H <sub>2</sub>                                                           | -0.41                                 |

reduction; therefore, the other reactions are more favorable.<sup>88</sup> Moreover, the thermodynamic potential of CO<sub>2</sub> to CH<sub>4</sub> is more thermodynamically feasible than that required to reduce H<sup>+</sup> to H<sub>2</sub>. Therefore, the principle for the photocatalytic reduction of CO<sub>2</sub> toward fuels is relatively complex, and we assume that this process involves multiple proton-coupled electron transfers instead of a single electron transfer.<sup>89</sup> With the exception of the listed thermodynamic considerations, we also need to consider the kinetic challenges in the photocatalytic reduction of CO<sub>2</sub>.<sup>90</sup> From the perspective of kinetics, H<sub>2</sub> formation from the reduction of H<sub>2</sub>O is more favorable, which is a strong competing reaction when compared with CO<sub>2</sub> reduction. We often use selectivity (ratio of the number of electrons reacting with CO<sub>2</sub> reduction to the number of all the reaction electrons) to estimate the efficiency of CO<sub>2</sub> reduction. Therefore, the selectivity for CO<sub>2</sub> reduction is low.

### 2.2 Reaction mechanism

It is well known that the energy band of a semiconductor is divided into conduction band (CB) and valence band (VB). In order to achieve the photocatalytic reduction of CO<sub>2</sub>, a good photocatalyst should have suitable band positions and bandgap energy ( $E_g$ ). Namely, the position of CB should be higher or more negative than the redox potential for CO<sub>2</sub> reduction, whereas the VB should be lower or more positive than the oxidation potential of H<sub>2</sub>O to O<sub>2</sub>,<sup>91,92</sup> and the absorbed light energy ( $h\nu$ ) should be greater than or equal to  $E_g$ .<sup>93,94</sup> Fig. 1 shows the band positions of some typical semiconductors, such as TiO<sub>2</sub>, Cu<sub>2</sub>O, ZnO, CdS, and Bi<sub>2</sub>WO<sub>6</sub>.<sup>95–101</sup>

The typical processes for the photocatalytic reduction of CO<sub>2</sub> are described in Fig. 2.<sup>101,102</sup> The process involves the following three main steps. First, if  $h\nu$  is greater than or equal to  $E_g$  under light irradiation, e<sup>-</sup> and h<sup>+</sup> can be produced. Considering the visible light utilization in solar energy, we should develop visible light response photocatalysts that have narrow bandgap energies ( $E_g < 3.0$  eV or  $\lambda > 415$  nm).<sup>103,104</sup> Second, e<sup>-</sup> and h<sup>+</sup> migrate toward the surface of the photocatalyst; at the same time, most e<sup>-</sup> and h<sup>+</sup> may get consumed through the recombination in the surface and bulk. In order to obtain high overall efficiency, it is necessary to quickly transfer the e<sup>-</sup>/h<sup>+</sup> pairs to the surface and suppress the recombination of e<sup>-</sup>/h<sup>+</sup> pairs. Third, after migrating toward the surface of the photocatalyst,

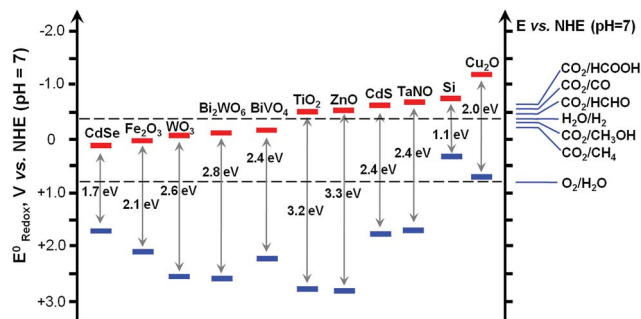


Fig. 1 Band positions of some semiconductors relative to the energy levels of CO<sub>2</sub> reduction. Reproduced from ref. 101 with permission from The Royal Society of Chemistry.

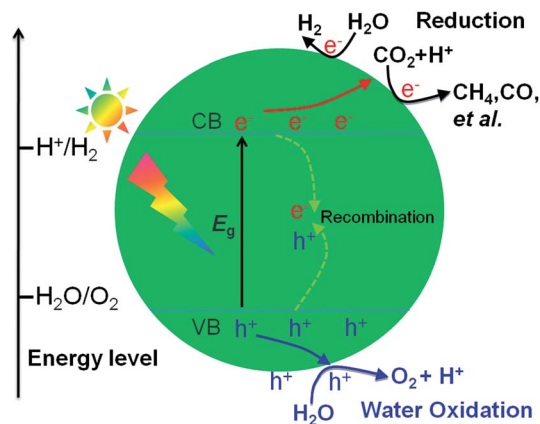


Fig. 2 Typical process of the photocatalytic reduction of CO<sub>2</sub>. Reproduced from ref. 101 with permission from The Royal Society of Chemistry.

$e^-$  and  $h^+$  are used for the reduction and oxidation reactions, respectively. Here,  $e^-$  reacts with CO<sub>2</sub> and  $H^+$  to produce CO and hydrocarbon fuels or only with  $H^+$  to produce H<sub>2</sub>. Further,  $h^+$  reacts with H<sub>2</sub>O to produce O<sub>2</sub>, thereby accelerating H<sub>2</sub>O oxidation and improving the separation of  $e^-/h^+$  pairs to further enhance the activity. Increasing the activity of CO<sub>2</sub> reduction is a formidable challenge. The recombination of  $e^-/h^+$  pairs is the determinant factor for photocatalytic activity.<sup>105–109</sup> Therefore, in order to obtain high activity for CO<sub>2</sub> reduction, we should formulate certain effective strategies, such as using a heterojunction to increase the charge transfer and restrain the recombination of  $e^-/h^+$  pairs.<sup>110–115</sup>

### 3. Design and fabrication of a photoreactor

High photocatalytic activity is not only dependent on the catalyst but also on the photoreactor.<sup>116</sup> The reaction modes (solid-liquid and solid-vapor interfaces) and operation modes (batch, semi-batch, and continuous) are important parameters in the design and fabrication of a photoreactor. It is noteworthy that the operation parameters, *e.g.*, temperature, pressure, light

source, and light intensity, play important roles in yielding high photocatalytic performances.

#### 3.1 Reaction mode

The solid-liquid and solid-vapor interfaces are the two typical reaction modes for the photocatalytic reduction of CO<sub>2</sub>. The schematic of two typical reaction modes are shown in Fig. 3. Fujishima and Honda conducted the pioneering work in the photocatalytic reduction of CO<sub>2</sub> at the solid-liquid interface,<sup>21</sup> and many researchers have since used this reaction mode.<sup>117</sup> The solid-liquid interface reaction mode is very simple, where the catalyst is dispersed in a solution. However, the solubility of CO<sub>2</sub> in water is very low, resulting in low photocatalytic activity. In order to overcome this issue, an alkaline medium has been used in a solution to increase the adsorption of CO<sub>2</sub>. Nonetheless, CO<sub>3</sub><sup>2-</sup> and HCO<sub>3</sub><sup>-</sup> can be formed in an alkaline medium. Furthermore, CO<sub>3</sub><sup>2-</sup> and HCO<sub>3</sub><sup>-</sup> are difficult to reduce as compared to CO<sub>2</sub>.<sup>118</sup> The solid-vapor interface reaction mode can circumvent these problems.<sup>119</sup> Wang *et al.* compared the solid-liquid interface and solid-vapor interface reaction mode of 0.5 wt% Pt-TiO<sub>2</sub> in the photocatalytic reduction of CO<sub>2</sub>.<sup>61,101</sup> It was interesting to find that the CH<sub>4</sub> formation rate was 5.2 μmol g<sup>-1</sup> h<sup>-1</sup> in the solid-vapor interface, which was 3.7 times higher than that of the solid-liquid interface. However, the H<sub>2</sub> formation rate in the solid-vapor interface was lower than that of the solid-liquid interface. In the solid-vapor interface, the catalyst was exposed in a CO<sub>2</sub> atmosphere to avoid H<sub>2</sub> formation. Therefore, the selectivity for CO<sub>2</sub> reduction was much higher in the solid-vapor interface, which was more suitable for CO<sub>2</sub> reduction in the presence of H<sub>2</sub>O.

#### 3.2 A typical photoreactor

**3.2.1 Slurry photoreactor.** In Fig. 4a, a slurry photoreactor with a suspended catalyst is shown, which has been widely used in the photocatalytic reduction of CO<sub>2</sub>.<sup>120</sup> It has some advantages, such as simple construction, high amount of catalyst, and low cost. Furthermore, it is possible to design and fabricate a large-capacity photoreactor. However, catalyst separation and circulation are drawbacks. Bai *et al.* first reported Mo-doped titanate nanotube (TNT) photocatalysts in a monoethanolamine solution with a slurry batch photoreactor under UVA (8 W, 63 mW cm<sup>-2</sup>) irradiation for the photocatalytic reduction of CO<sub>2</sub> into CH<sub>4</sub> and CO.<sup>121</sup> It was clearly found that the molybdenum structure and oxygen vacancies played important roles in enhancing the photocatalytic performance. Maroto-Valer *et al.* fabricated 1 wt% Pd/0.01 wt% Rh-TiO<sub>2</sub> in a slurry batch annular reactor system under UVA irradiation.<sup>122</sup> The obtained quantum efficiency was about 0.002% and CH<sub>4</sub> formation rate was 0.03 μmol g<sup>-1</sup> h<sup>-1</sup>. In order to realize the separation of H<sub>2</sub> and O<sub>2</sub>, Wu *et al.* designed a slurry reactor with two separate components for O<sub>2</sub> and H<sub>2</sub> evolution through a Nafion membrane (Fig. 4b).<sup>123</sup> Pt/CuAlGaO<sub>4</sub> or Pt/SrTiO<sub>3</sub>:Rh served as the catalyst in CO<sub>2</sub> reduction to produce CO, and WO<sub>3</sub> was used as the oxidation catalyst for the oxidation of H<sub>2</sub>O to produce O<sub>2</sub>. This dual-photocatalyst system yielded high

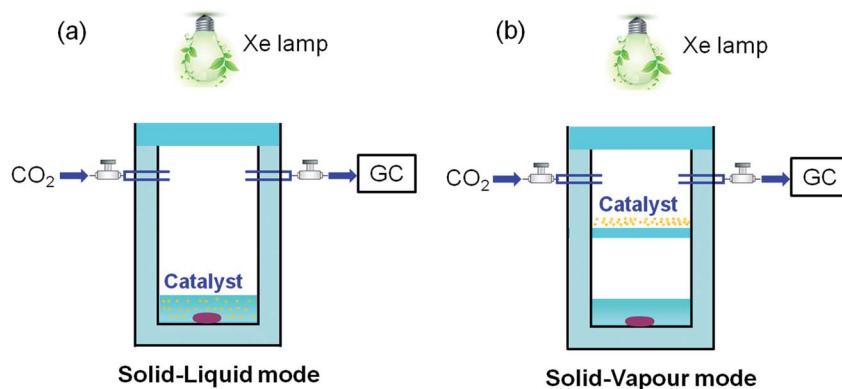


Fig. 3 Two typical reaction modes for the photocatalytic reduction of  $\text{CO}_2$ . (a) Solid-liquid interface reaction mode; (b) solid-vapor interface reaction mode. (a and b) Reproduced from ref. 61 with permission from the American Chemical Society and from ref. 101 with permission from The Royal Society of Chemistry.

quantum efficiency (0.0051%), which was 2.7 times higher than that of a single-photocatalyst system (0.0019%).

**3.2.2 Optical-fiber photoreactor.** A fixed-bed photoreactor can overcome the disadvantages of catalyst separation and circulation encountered in a slurry photoreactor. The catalysts are often coated onto the fixed supports, such as plates, beads, fibers, and the monolith.<sup>5</sup> The fibers can be used as a medium to uniformly deliver light onto the surface of the catalyst. In the optical-fiber photoreactor, the catalyst coated onto the fibers can direct light to enter along the fiber.<sup>120</sup> Fig. 5a shows the photograph of an optical-fiber photoreactor.<sup>124</sup> It has certain advantages such as high surface area and light utilization efficiency; however, it has a comparatively low surface area and low reactor volume with respect to the photocatalysis process. The  $\text{NiO}/\text{InTaO}_4$  catalyst was prepared *via* the sol-gel method and then coated onto optical fibers and calcined to obtain a uniform layer.<sup>124</sup> The quantum efficiency of the optical-fiber reactor was 0.063%, which was much higher than that of the aqueous-phase reactor (0.0045%). In a similar work, Wu *et al.* also prepared  $\text{Cu}/\text{TiO}_2$  for the photocatalytic reduction of  $\text{CO}_2$  in an optical-fiber photoreactor, which showed the maximum  $\text{CH}_3\text{OH}$  formation rate ( $0.45 \mu\text{mol g}^{-1} \text{h}^{-1}$ ) under 365 nm wavelength irradiation.<sup>125</sup> Wu *et al.* also reported metal-loaded  $\text{TiO}_2$  films coated onto fibers in an optical-fiber photoreactor.<sup>126</sup> It was found that

the maximum quantum efficiency was obtained with 1.2% Cu under 365 nm UV irradiation. In a further study, Wu *et al.* also researched the  $\text{Cu-Fe}/\text{TiO}_2$  coated onto fibers; Fe and  $\text{Cu-TiO}_2$  simultaneously promoted  $\text{C}_2\text{H}_4$  formation and the quantum yield was about 0.024%.<sup>127</sup> A novel optical-fiber monolith reactor with multiple inverse lights for the photocatalytic reduction of  $\text{CO}_2$  to produce  $\text{CH}_3\text{OH}$  was reported by Yang *et al.*<sup>128</sup> The maximum quantum efficiency was 0.0177%, which was higher than the internally illuminated monolith.

**3.2.3 Monolith photoreactor.** The monolith photoreactor has high surface area, low pressure drop, and ease in scale-up. The monolith is made up of many internal channels because of which it has a large surface area.<sup>129</sup> The catalyst is coated onto the walls of the channels. However, the light cannot efficiently penetrate through the channels because of the opacity of the honeycomb substrate; therefore, the length of the monolith reactor is limited. Fig. 5b shows the photograph of a monolith photoreactor.<sup>130</sup> Wu *et al.* reported that  $\text{NiO}/\text{InTaO}_4$  was coated onto a pre-coated  $\text{SiO}_2$  sublayer on the internal channels of the monolith for the photocatalytic reduction of  $\text{CO}_2$  in a monolith photoreactor.<sup>130</sup> The maximum  $\text{CH}_3\text{OH}$  formation rate was  $0.16 \mu\text{mol g}^{-1} \text{h}^{-1}$  under visible light irradiation. Moreover, the quantum efficiency of the monolith photoreactor was 0.012%, which was much higher than that of the optical-fiber reactor.

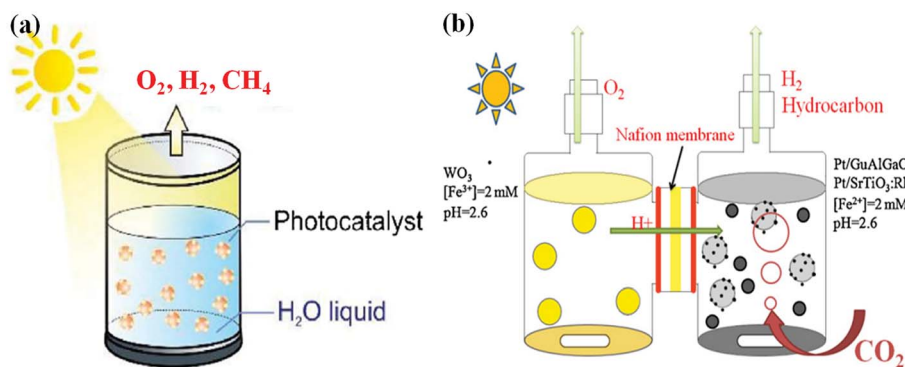


Fig. 4 Schematic diagram of (a) a typical slurry photoreactor and (b) a novel twin reactor system. (a) Reproduced from ref. 120 with permission from Elsevier. (b) Reproduced from ref. 123 with permission from Elsevier.

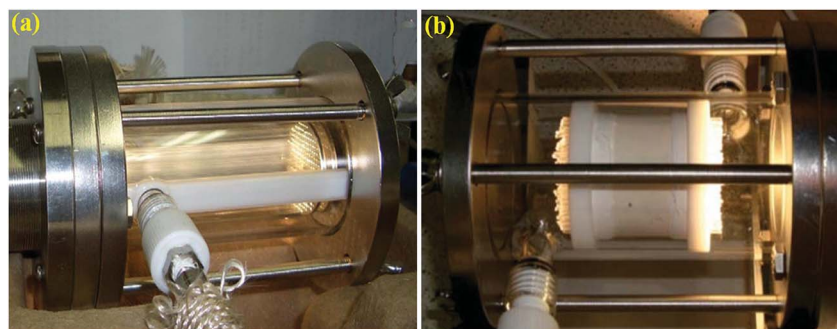


Fig. 5 Photographs of (a) an optical-fiber photoreactor and (b) a monolith photoreactor. (a) Reproduced from ref. 124 with permission from Elsevier. (b) Reproduced from ref. 130 with permission from The Royal Society of Chemistry.

Amin *et al.* reported montmorillonite (MMT)/TiO<sub>2</sub> for CO<sub>2</sub> reduction in a monolith photoreactor.<sup>131</sup> It exhibited a high CH<sub>4</sub> formation rate (139 μmol g<sup>-1</sup> h<sup>-1</sup>) when compared with a cell-type photoreactor (43 μmol g<sup>-1</sup> h<sup>-1</sup>) owing to the high illuminated surface area and light utilization. Tahir *et al.* developed a Cu- and N-doped TiO<sub>2</sub> nanocatalyst coated onto honeycomb monoliths for CO<sub>2</sub> reduction in a monolith photoreactor.<sup>132</sup> It was noteworthy that Cu and N could increase the activity and selectivity for CO<sub>2</sub> reduction to yield CO.

**3.2.4 Optofluidic microreactor.** An optofluidic microreactor has a precise control of flow, large surface area, enhanced mass transfer, and uniform light distribution; however, there exists a severe problem of limited throughput for practical applications. Fig. 6a and b show the schematic and photograph of an optofluidic planar microreactor, respectively.<sup>133</sup> Zhu *et al.* used an

optofluidic planar microreactor to enhance the photocatalytic activity of a TiO<sub>2</sub> film in an alkaline environment.<sup>133</sup> The increase in the light intensity and NaOH concentration could increase the CH<sub>3</sub>OH yields, and the increase in the liquid flow and catalyst loading initially improved and then decreased for the CH<sub>3</sub>OH yields. In a similar study, Cheng *et al.* reported Cu-TiO<sub>2</sub> nanorods for the photocatalytic reduction of CO<sub>2</sub> in an optofluidic planar microreactor.<sup>134</sup> It had the maximum CH<sub>3</sub>OH and CH<sub>3</sub>CH<sub>2</sub>OH yields with a flow rate of 2 mL min<sup>-1</sup>. Fig. 6c and d show the schematic and photograph of an optofluidic membrane microreactor.<sup>135</sup> An optofluid with the membrane reactor technology for CO<sub>2</sub> reduction of TiO<sub>2</sub>/carbon paper was reported by Chen *et al.*<sup>135</sup> The CH<sub>3</sub>OH yield was much higher than those obtained from other types of photoreactors.

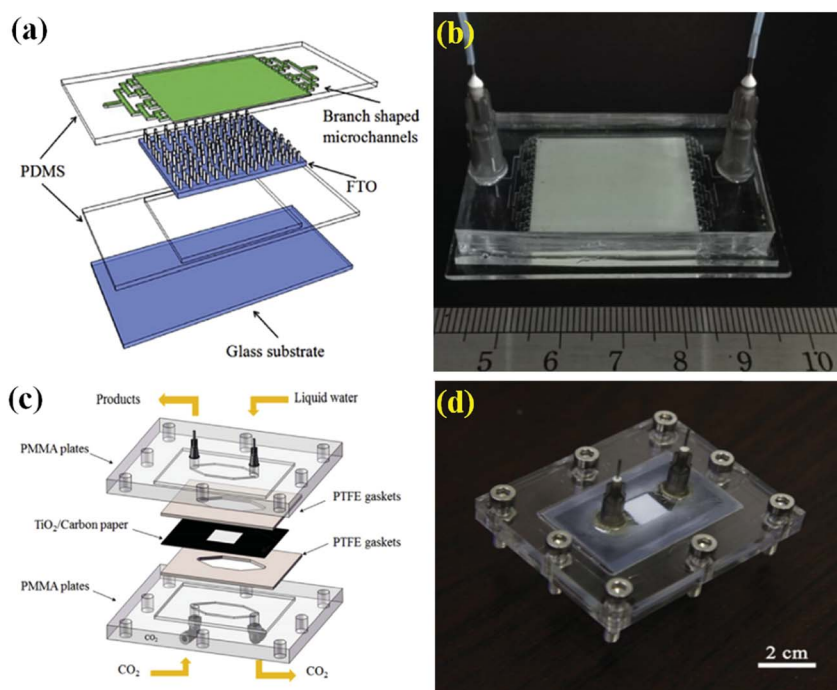


Fig. 6 Designs and photographs of (a and b) an optofluidic planar microreactor and (c and d) an optofluidic membrane microreactor. (a and b) Reproduced from ref. 134 with permission from Elsevier. (c and d) Reproduced from ref. 135 with permission from Elsevier.

## 4. Design and fabrication of TiO<sub>2</sub>-based heterojunction photocatalysts

### 4.1 Coupling with other semiconductors

With regard to bare semiconductor photocatalysts, many strategies have been used to enhance their photocatalytic performance. A well-defined heterojunction structure between two semiconductors can accelerate the charge separation and suppress the recombination of e<sup>-</sup>/h<sup>+</sup> pairs. A p-n heterojunction can be formed when n-type TiO<sub>2</sub> couples with some appropriate p-type semiconductor with matching electronic band structures. Fig. 7a shows the schematic diagram of the energy band structure and e<sup>-</sup>/h<sup>+</sup> pair separation of a p-n heterojunction.<sup>136</sup> When n-type TiO<sub>2</sub> contacts with a p-type

semiconductor, the diffusion of e<sup>-</sup> and h<sup>+</sup> can form a space charge region at the interface. The difference in the electric potential can create a strong electric field to accelerate the charge separation of e<sup>-</sup>/h<sup>+</sup> pairs. Except for p-n heterojunctions, other non-p-n heterojunctions exist. The schematic diagram of the energy band structure and e<sup>-</sup>/h<sup>+</sup> pair separation of the non-p-n heterojunction are shown in Fig. 7b. Semiconductor A (SA) and semiconductor B (SB) are tightly bonded to form a heterojunction. Owing to the staggered alignment of the energy levels, e<sup>-</sup> can transfer from the CB of SA to the CB of SB, whereas h<sup>+</sup> transfers from the VB of SB to the VB of SA under light irradiation. The non-p-n heterojunction is also suitable for the photocatalytic reduction of CO<sub>2</sub> into solar fuels, and TiO<sub>2</sub> is often used to fabricate such heterojunctions.

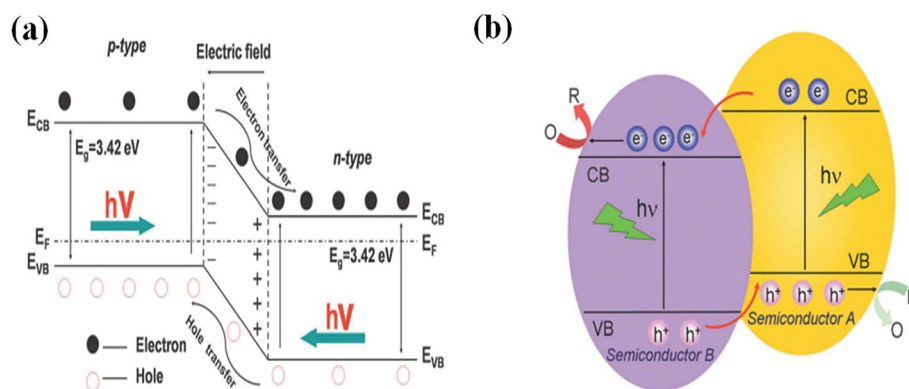


Fig. 7 Schematic diagram showing the energy band structure and e<sup>-</sup>/h<sup>+</sup> pair separation in p-n (a) and non-p-n (b) heterojunctions. Reproduced from ref. 136 with permission from The Royal Society of Chemistry.

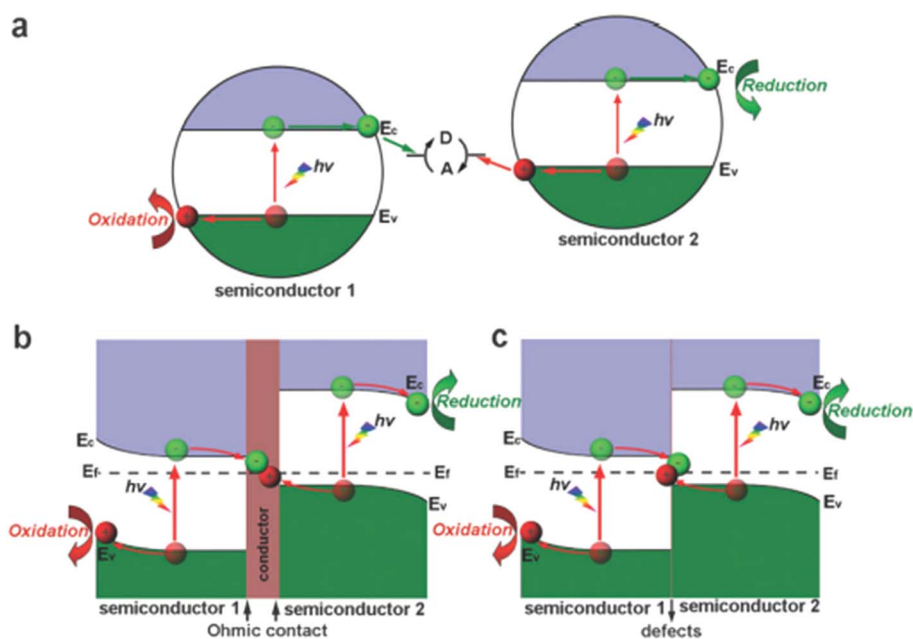


Fig. 8 Schematics of the (a) indirect Z-scheme structure mediated by redox pairs; (b) semiconductor-conductor-semiconductor all-solid-state Z-scheme structure; (c) semiconductor-semiconductor all-solid-state Z-scheme structure. Reproduced from ref. 138 with permission from The Royal Society of Chemistry.

Recently, another Z-scheme heterojunction system has aroused considerable interest.<sup>137</sup> It is typically made up of two semiconductors with staggered alignment of band structures. However, the CB and VB of the two semiconductors are only able to initiate reduction or oxidation of half of the reaction and unable to individually sustain the overall reactions. Fig. 8a shows the indirect Z-scheme heterojunction structure;  $e^-$  in the CB of semiconductor 1 recombines with  $h^+$  in the VB of semiconductor 2;  $e^-$  in the CB of semiconductor 2 and  $h^+$  in the VB of semiconductor 1 are utilized for the reduction and oxidation reactions, respectively.<sup>138,139</sup> Nevertheless, this structure can decrease the amount of  $e^-$  and  $h^+$  available for the reactions and lead to low photocatalytic activity. The all-solid-state Z-scheme heterojunction structure can circumvent these problems. Fig. 8b and c show that a conductor serves as the electron

mediator and many defects exist at the solid–solid interface. Recently, a novel double  $Zn_3(VO_4)_2/Zn_2V_2O_7/ZnO$  Z-scheme ternary vanadate heterojunction has been synthesized by Yu *et al.*, which exhibited high photocatalytic activity due to fast interfacial charge transfer and wider light response.<sup>140</sup> Table 3 summarizes and compares the typical  $TiO_2$ –semiconductor heterojunction photocatalysts for the photocatalytic reduction of  $CO_2$ .

**4.1.1 p–n heterojunction.** When  $TiO_2$  and some p-type semiconductor are in contact, a strong electric field can be created. The produced electric field can accelerate charge separation and suppress the recombination of  $e^-/h^+$  pairs. It is well known that copper compounds are very promising materials for the photocatalytic reduction of  $CO_2$ , and they are usually used to fabricate such p–n heterojunctions.

Table 3 Comparison of typical  $TiO_2$ –semiconductor heterojunction systems

| Photocatalyst                       | Solvent             | Light source                     | Time <sup>a</sup> | Formation rate <sup>b</sup>                                           | QE <sup>c</sup> (%)          | Reference |
|-------------------------------------|---------------------|----------------------------------|-------------------|-----------------------------------------------------------------------|------------------------------|-----------|
| <b>p–n heterojunction</b>           |                     |                                  |                   |                                                                       |                              |           |
| $Cu_2O/TiO_2$ porous                | $H_2O$              | Xe lamp ( $\lambda > 300$ nm)    | 10                | $CH_4$ : 0.0284                                                       | —                            | 145       |
| $Cu_2O/TiO_2$ hollow nanospheres    | $H_2O$              | Xe lamp ( $\lambda \geq 420$ nm) | 8                 | $CH_4$ : 0.16                                                         | —                            | 146       |
| $Cu_2O/TiO_2$                       | $H_2O$              | UVA                              | 1.5               | $CH_4$ : 0.024                                                        | —                            | 147       |
| $TiO_2/Ti_4O_9/Cu_2O$               | $H_2O$              | AM 1.5                           | 5                 | $CH_3OH$ : 0.586                                                      | —                            | 148       |
| $Cu_2O/TiO_2$ nanotube              | $H_2O$              | Xe Hg                            | 6                 | $CH_3OH$ : $9.19 \mu M mL^{-1} h^{-1}$                                | 1.731 <sup>d</sup>           | 149       |
| $Cu_2O/TiO_2$ nanotube              | $H_2O$              | Xe lamp ( $\lambda < 420$ nm)    | 4                 | $CH_4$ : $2 ppm h^{-1}$                                               | —                            | 150       |
| $CuO/TiO_2$ nanotube                | $H_2O$              | Hg lamp                          | 3                 | $CH_4 + CH_3OH$                                                       | —                            | 154       |
| $CuO/TiO_2$                         | $CH_3OH$            | Hg lamp (365 nm)                 | 6                 | $HCOOCH_3$ : 1600                                                     | —                            | 157       |
| $CuO/TiO_2$                         | 1 M $KHCO_3$        | UV lamps                         | 6                 | $CH_3OH$ : 442.5                                                      | 19.23                        | 158       |
| $CuO/TiO_2$ hollow microspheres     | $H_2O$              | Hg UV lamp                       | 24                | $CO$ : 5.47; $CH_4$ : 0.36                                            | $CO$ : 1.285; $CH_4$ : 0.747 | 159       |
| Hybrid $Cu_xO-TiO_2$ ( $x = 1, 2$ ) | $H_2O$              | AM 1.5                           | 1                 | $CH_4$ : 0.22163                                                      | —                            | 160       |
| $CuO-TiO_{2-x}N_x$                  | $H_2O$              | AM 1.5                           | 3                 | $CH_4$ : $41.3 ppm g^{-1} h^{-1}$                                     | —                            | 161       |
| $GaP/TiO_2$                         | $H_2O$              | Xe lamp                          | 24                | $CH_4$ : 2.46                                                         | —                            | 163       |
| <b>Non p–n heterojunction</b>       |                     |                                  |                   |                                                                       |                              |           |
| $CdS-TiO_2$                         | Cyclohexanol        | Xe lamp                          | 10                | Cyclohexyl formate: 20.2                                              | —                            | 169       |
| $CdS-TiO_2$                         | 2 M $NaNO_2$        | Hg lamp                          | 8                 | $CO$ : 2; $CH_4$ : 0.375                                              | —                            | 171       |
| $CdS-TiO_2$                         | 2 M $NaNO_2$        | Hg lamp (UV-visible)             | 8                 | $CO$ : 1.3; $CH_4$ : 0.1975                                           | —                            | 171       |
| $CdS-TiO_2$ nanotube                | 2 M $NaNO_2$        | Hg lamp (visible)                | 5                 | $CH_3OH$ : 31.9                                                       | —                            | 172       |
| $Bi_2S_3-TiO_2$ nanotube            | 2 M $NaNO_2$        | Xe lamp                          | 5                 | $CH_3OH$ : 44.9                                                       | —                            | 172       |
| $PbS QDs-TiO_2$                     | $H_2O$              | Xe lamp                          | 8                 | $CH_4$ : 0.58; $CO$ : 0.82; $C_2H_6$ : 0.31                           | —                            | 173       |
| $CeO_2-TiO_2$                       | $H_2O$              | Xe lamp                          | 6                 | $CO$ : 3.09                                                           | —                            | 174       |
| $CeO_2-TiO_2$                       | $H_2O$              | Xe lamp                          | 6                 | $CH_4$ : 2200; $CO$ : 13 600                                          | —                            | 175       |
| $TiO_2/ZnO$                         | $H_2O$              | Xe lamp                          | 5                 | $CH_4$ : 55                                                           | —                            | 177       |
| V and W doped $TiO_2$               | $H_2O$              | Xe lamp                          | 4                 | $CH_4$ : 0.22; $CO$ : 1.91                                            | 0.063                        | 178       |
| $FeTiO_3/TiO_2$                     | 2 M $NaNO_2$        | Xe lamp ( $\lambda > 300$ nm)    | 3                 | $CH_3OH$ : 0.462                                                      | —                            | 179       |
| $FeTiO_3/TiO_2$                     | 2 M $NaNO_2$        | Xe lamp ( $\lambda > 400$ nm)    | 3                 | $CH_3OH$ : 0.432                                                      | —                            | 179       |
| $Au_3Cu@SrTiO_3/TiO_2$              | $N_2H_4 \cdot H_2O$ | Xe lamp                          | 6                 | $CO$ : 3770; $CH_4$ : 421.2                                           | 2.51                         | 180       |
| $TiO_2/g-C_3N_4$                    | $H_2O$              | Hg lamp                          | 8                 | $CH_4$ : 8.8; $CO$ : 2.9                                              | —                            | 181       |
| $g-C_3N_4-N-TiO_2$                  | $H_2O$              | Xe lamp                          | 12                | $CO$ : 12.28                                                          | —                            | 182       |
| $AgBr/TiO_2$                        | 0.2 M $KHCO_3$      | Xe lamp                          | 5                 | $CH_4$ : 128.56; $CH_3OH$ : 77.87; $CH_3CH_2OH$ : 13.28; $CO$ : 32.14 | —                            | 183       |
| <b>Z-scheme heterojunction</b>      |                     |                                  |                   |                                                                       |                              |           |
| $Si/TiO_2$                          | $H_2O$              | Xe lamp                          | 3                 | $CH_3OH$                                                              | 18.1 <sup>d</sup>            | 186       |
| $ZnFe_2O_4/TiO_2$                   | Cyclohexanol        | Hg lamp                          | 8                 | Cyclohexyl formate: 22.26                                             | —                            | 187       |
| $CdS/rGO/TiO_2$                     | $H_2O$              | Xe lamp                          | 10                | $CH_4$ : 0.12                                                         | —                            | 188       |

<sup>a</sup> Irradiation time, h. <sup>b</sup> Formation rate,  $\mu mol g^{-1} h^{-1}$ . <sup>c</sup> Quantum efficiency (%). <sup>d</sup> Photonic efficiency (%).



Furthermore, copper compounds can improve the selectivity for CO<sub>2</sub> reduction.<sup>141</sup>

Cuprous oxide (Cu<sub>2</sub>O) is a p-type semiconductor with a narrow direct bandgap (~2.0 eV), which is a promising material for CO<sub>2</sub> reduction.<sup>142–144</sup> Ye *et al.* synthesized a porous-structured Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction *via* a two-step process as follows.<sup>145</sup> First, titanium tetrachloride (TiCl<sub>4</sub>) was used as the Ti precursor and heated with microwave assistance at 200 °C for 40 min; then, porous TiO<sub>2</sub> was synthesized and dispersed into ethanol. Second, copper(II) acetylacetonate (Cu(acac)<sub>2</sub>) was used as the Cu precursor and added into TiO<sub>2</sub> dispersions and treated with a microwave-solvothermal process at 200 °C for 40 min. Then, a porous Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction was obtained. The BET surface area of Cu<sub>2</sub>O/TiO<sub>2</sub> (206.3 m<sup>2</sup> g<sup>-1</sup>) was 4.04 and 1.04 times higher than that of P25 (51.1 m<sup>2</sup> g<sup>-1</sup>) and porous TiO<sub>2</sub> (198.2 m<sup>2</sup> g<sup>-1</sup>), respectively. Moreover, the amounts of adsorbed CO<sub>2</sub> of Cu<sub>2</sub>O/TiO<sub>2</sub> (0.27 mL g<sup>-1</sup>) was much higher than that of P25 (0.05 mL g<sup>-1</sup>) and porous TiO<sub>2</sub> (0.12 mL g<sup>-1</sup>). The photocatalytic performance was evaluated by the photocatalytic reduction of CO<sub>2</sub> under UV-vis light irradiation ( $\lambda > 300$  nm). It was found that the highest CH<sub>4</sub> formation rate could be achieved by the Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction catalyst. Fig. 9 shows the schematic of the charge transfer in the Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction. The bandgaps of TiO<sub>2</sub> and Cu<sub>2</sub>O were 3.2 and 2.1 eV, respectively. Under light irradiation, e<sup>-</sup> could transfer from the CB of Cu<sub>2</sub>O to the CB of TiO<sub>2</sub>, whereas h<sup>+</sup> could transfer from the VB of TiO<sub>2</sub> to the VB of Cu<sub>2</sub>O, resulting in promoted separation efficiency of the photogenerated e<sup>-</sup> and h<sup>+</sup>. This superior performance was attributed to the heterojunction with high separation efficiency of e<sup>-</sup>/h<sup>+</sup> pairs, as well as high CO<sub>2</sub> adsorption and more active reaction sites in the porous structure.

Morphology is expected to play a unique role in the photocatalytic performance. In a previous study, He *et al.* first fabricated Cu<sub>2</sub>O hollow nanospheres by a soft-template method and then loaded them onto TiO<sub>2</sub> through the *in situ* hydrolysis of Ti(OBu)<sub>4</sub> under ultrasonic conditions.<sup>146</sup> Fig. 10 shows the SEM

and TEM images of Cu<sub>2</sub>O and Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction. The diameter of Cu<sub>2</sub>O hollow nanospheres was about 80–120 nm, and the Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction also retained the same hollow spheres and diameter. In this case, the highest CH<sub>4</sub> formation rate was obtained by the Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction under visible light irradiation. Cu<sub>2</sub>O in contact with TiO<sub>2</sub> could form a p–n heterojunction to yield the efficient separation of photoinduced charge carriers as well as stability. In another study, Wu *et al.* reported Cu dispersion over Cu<sub>2</sub>O/TiO<sub>2</sub> for the photocatalytic reduction of CO<sub>2</sub> using a stirred batch annular reactor under UVA irradiation.<sup>147</sup> A volcano tendency was evident in the CH<sub>4</sub> yield loading with moderate Cu amounts. It was found that the largest CH<sub>4</sub> yield could be obtained with Cu loading amount of 0.03%. Recently, a novel example of multi-component pillared structures of Cu<sub>2</sub>O-decorated TiO<sub>2</sub>-pillared tetratitanate was synthesized by Nogueira *et al.*<sup>148</sup> The BET surface area of TiO<sub>2</sub>/Ti<sub>4</sub>O<sub>9</sub>/Cu<sub>2</sub>O (171 m<sup>2</sup> g<sup>-1</sup>) was 6.84 and 1.18 times higher than that of K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub> (25 m<sup>2</sup> g<sup>-1</sup>) and TiO<sub>2</sub>/Ti<sub>4</sub>O<sub>9</sub> (145 m<sup>2</sup> g<sup>-1</sup>), respectively. TiO<sub>2</sub>/Ti<sub>4</sub>O<sub>9</sub>/Cu<sub>2</sub>O exhibited the maximum CH<sub>3</sub>OH formation rate, which was attributed to the synergistic effect induced by the pillaring K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub> with TiO<sub>2</sub> pillars and the loading of the Cu<sub>2</sub>O cocatalyst; this yielded the heterojunction structure. These three components could generate e<sup>-</sup> and h<sup>+</sup> under light irradiation. However, Cu<sub>2</sub>O was the main photocatalyst to generate e<sup>-</sup> and h<sup>+</sup> owing to its narrowest bandgap. Further, e<sup>-</sup> in the CB of TiO<sub>2</sub> could transfer to the CB of H<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub>, which then coupled with h<sup>+</sup> in the VB of Cu<sub>2</sub>O. Furthermore, e<sup>-</sup> in the CB of Cu<sub>2</sub>O reacted with the surface-adsorbed CO<sub>2</sub> and H<sup>+</sup> to produce CH<sub>3</sub>OH and h<sup>+</sup> in the VB of TiO<sub>2</sub> reacted with H<sub>2</sub>O to produce O<sub>2</sub> and H<sup>+</sup>. The electric field at the heterojunction interfaces could accelerate the transfer of e<sup>-</sup>/h<sup>+</sup> pairs and the onset of the absorption band of the semiconductor exhibited a red-shift to improve the CH<sub>3</sub>OH formation rate.

Ji *et al.* used the electrodeposition method to deposit Cu<sub>2</sub>O nanoparticles (NPs) onto TiO<sub>2</sub> nanotubes (NTs) to fabricate the

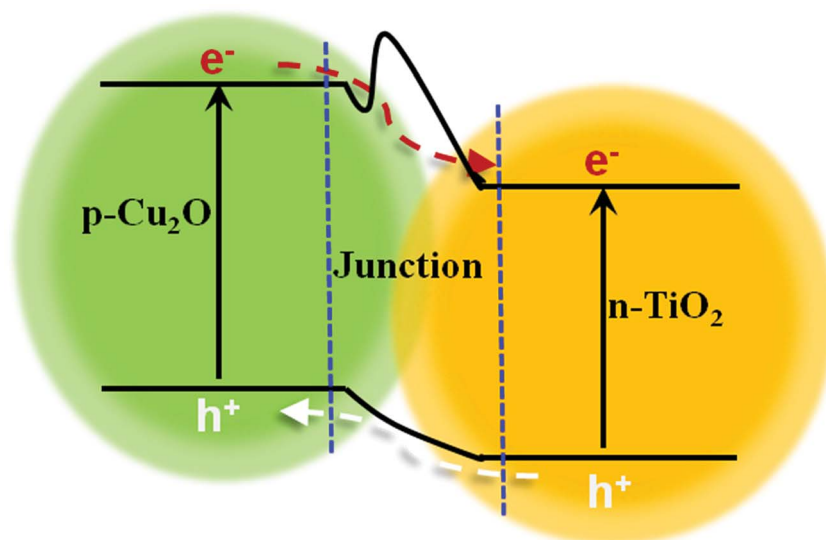


Fig. 9 Schematic of the charge transfer process of Cu<sub>2</sub>O/TiO<sub>2</sub> heterojunction.

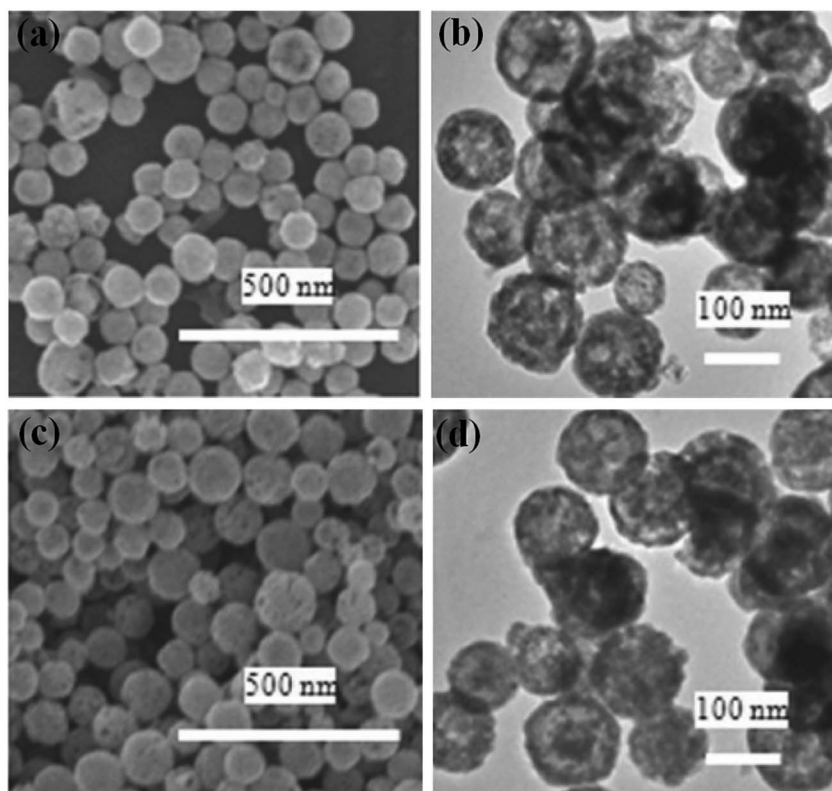


Fig. 10 SEM and TEM images of (a and b)  $\text{Cu}_2\text{O}$  and (c and d)  $\text{Cu}_2\text{O}/\text{TiO}_2$  heterojunctions. Reproduced from ref. 146 with permission from Wiley-VCH.

$\text{Cu}_2\text{O}/\text{TiO}_2$  NT heterostructure.<sup>149</sup> Fig. 11a and b show the SEM images of the  $\text{TiO}_2$  NTs. The average inner diameter and length of  $\text{TiO}_2$  NTs were about 80 nm and 800 nm, respectively. Fig. 11c and d show the top and cross-sectional views of the  $\text{Cu}_2\text{O}/\text{TiO}_2$  NT heterostructure.  $\text{Cu}_2\text{O}$  NPs with a polyhedral shape were formed on the surface of the  $\text{TiO}_2$  NTs. Moreover, the  $\text{Cu}_2\text{O}$  NPs were wrapped around the inner and outer walls of the  $\text{TiO}_2$  NTs. The  $\text{Cu}_2\text{O}/\text{TiO}_2$  NTs exhibited stronger adsorption in the UV and visible light regions than that of  $\text{TiO}_2$  NTs. The  $\text{Cu}_2\text{O}/\text{TiO}_2$  NTs exhibited good photocatalytic activity for the photocatalytic reduction of  $\text{CO}_2$  to  $\text{CH}_3\text{OH}$ , whereas the  $\text{TiO}_2$  NTs were nearly inactive. The band energy values and charge carrier transfers of  $\text{Cu}_2\text{O}/\text{TiO}_2$  NTs under visible light and UV-vis light irradiation are shown in Fig. 11e and f, respectively. It is well known that  $\text{TiO}_2$  is a UV response semiconductor; therefore, only  $e^-$  can be generated in the CB of  $\text{Cu}_2\text{O}$  under visible light irradiation, which then transfers to the CB of  $\text{TiO}_2$ . However, both  $\text{Cu}_2\text{O}$  and  $\text{TiO}_2$  can generate  $e^-$  under UV-vis light irradiation. Further,  $e^-$  in the CB of  $\text{Cu}_2\text{O}$  can transfer to the CB of  $\text{TiO}_2$ , whereas  $h^+$  in the VB of  $\text{TiO}_2$  can transfer to the VB of  $\text{Cu}_2\text{O}$ ; thereafter,  $e^-$  and  $h^+$  are effectively separated and the charge recombination process is suppressed. Due to the presence of  $\text{Cu}_2\text{O}/\text{TiO}_2$  NT heterostructures, the photocatalytic activity was improved. In another report, Yu *et al.* fabricated the octahedral  $\text{Cu}_2\text{O}$  NPs deposited onto  $\text{TiO}_2$  NTs by an electrochemical deposition method.<sup>150</sup> The largest  $\text{CH}_4$  production rate was 2 ppm  $\text{h}^{-1}$  under visible light irradiation with the deposition time for

30 min. The  $\text{TiO}_2$  NTs structure might offer high surface area and additional active sites. Furthermore, the  $\text{Cu}_2\text{O}$  NPs increased light absorption and enhanced the selectivity for  $\text{CO}_2$  reduction into hydrocarbon products.

Cupric oxide ( $\text{CuO}$ ) is a p-type semiconductor with a narrow direct bandgap, which is also a promising material for  $\text{CO}_2$  reduction.<sup>151–153</sup> Previous studies have clarified that  $\text{CuO}$  is a good semiconductor to form a heterojunction due to its ability to be used as an electron trap. When  $\text{CuO}$  couples with  $\text{TiO}_2$ , an electric field is created at the interface. Therefore, the charge separation of  $e^-$  and  $h^+$  is improved. Razali *et al.* reported  $\text{CuO}$  loaded onto  $\text{TiO}_2$  NTs by a facile hydrothermal method.<sup>154</sup> The diameter and length of  $\text{TiO}_2$  NTs with a fibrous-like structure were about 10 nm and several hundreds of nanometers, respectively. However, when  $\text{CuO}$  was loaded onto the  $\text{TiO}_2$  NT, the morphology exhibited a considerable change, where they were tied together at the tip and exhibited a bundled, cobweb-like appearance. In the photocatalytic reduction of  $\text{CO}_2$ , the  $\text{CuO}/\text{TiO}_2$  NT exhibited the highest  $\text{CO}_2$  conversion ( $\sim 100\%$  for 2.5 h), which was higher than other metal oxide composites, such as  $\text{ZnS}/\text{ZnO}$  (45%) and immobilized  $\text{MPC}/\text{ZnO}$  (23%).<sup>155,156</sup> The  $\text{CuO}/\text{TiO}_2$  NT heterojunction could improve the effective separation of  $e^-/h^+$  pairs, resulting in improved activity. In another study, Xin *et al.* reported the  $\text{CuO}-\text{TiO}_2$  photocatalyst for the photocatalytic reduction of  $\text{CO}_2$  into  $\text{HCOOCH}_3$  in the presence of  $\text{CH}_3\text{OH}$ , whereas  $\text{CH}_3\text{OH}$  was used as a sacrificial reagent to react with  $h^+$ .<sup>157</sup> The interfacial structure between

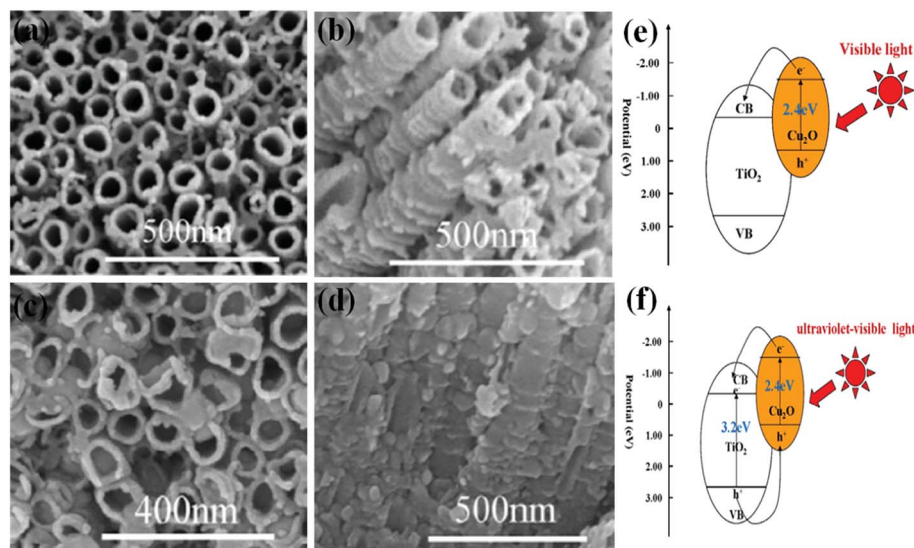


Fig. 11 SEM images of (a and b) TiO<sub>2</sub> NTs and (c and d) Cu<sub>2</sub>O/TiO<sub>2</sub> NT heterojunction; schematic of the charge separation of Cu<sub>2</sub>O/TiO<sub>2</sub> NT heterojunction structure under (e) visible light and (f) UV-vis light irradiations. Reproduced from ref. 149 with permission from Elsevier.

CuO and TiO<sub>2</sub> was characterized by HRTEM, and the HRTEM image is shown in Fig. 12a. The typical monoclinic CuO {110} and anatase TiO<sub>2</sub> {101} planes with the characteristic lattice spacing of 0.275 nm and 0.35 nm were observed, respectively. A surface-phase heterojunction existed between CuO and TiO<sub>2</sub>, and the band position of CuO and TiO<sub>2</sub> is shown in Fig. 12b. This heterojunction can improve charge migration and accelerate the separation of e<sup>-</sup>/h<sup>+</sup> pairs, enhancing the photocatalytic performance.

A series of Cu (Cu<sup>0</sup>, Cu<sup>I</sup>, Cu<sup>II</sup>)/TiO<sub>2</sub> photocatalysts *via* an improved impregnation method through the reduction-oxidation steps was fabricated by Gunlazuardi *et al.*<sup>158</sup> The 3 wt% CuO/TiO<sub>2</sub> exhibited the highest CH<sub>3</sub>OH yield (~442.5 μmol g<sup>-1</sup> h<sup>-1</sup>), which was 3.28, 2.28, and 1.97 times higher than TiO<sub>2</sub> (134.8 μmol g<sup>-1</sup> h<sup>-1</sup>), 3 wt% Cu/TiO<sub>2</sub> (194 μmol g<sup>-1</sup> h<sup>-1</sup>), and 3 wt% Cu<sub>2</sub>O/TiO<sub>2</sub> (224.3 μmol g<sup>-1</sup> h<sup>-1</sup>), respectively. It was clearly indicated that CuO was the main dopant species that

enhanced the CH<sub>3</sub>OH yield. In a further study, Wilkinson *et al.* also reported CuO loaded onto TiO<sub>2</sub> hollow microspheres by a one-pot template-free strategy.<sup>159</sup> Further, it exhibited a higher CH<sub>4</sub> formation rate than commercial Degussa P25 due to its large surface area, which contributed toward the adsorption of CO<sub>2</sub> and offered a large number of active sites.

In *et al.* synthesized hybrid Cu<sub>x</sub>O-TiO<sub>2</sub> (x = 1, 2) mesoporous heterojunction photocatalysts by two steps, and the synthesis process is shown in Fig. 13a.<sup>160</sup> Initially, hybrid mesoporous Cu<sub>x</sub>O/TiO<sub>2</sub> heterojunctions for the photocatalytic reduction of CO<sub>2</sub> into solar fuels were reported. CT07 (TiCl<sub>4</sub> was 0.7 mL mixed with the dispersion liquid of Cu/Cu<sub>2</sub>O nanocomposites) had a higher CH<sub>4</sub> formation rate (221.63 ppm g<sup>-1</sup> h<sup>-1</sup>) than CuO and TiO<sub>2</sub> due to the enhanced light absorption and rapid charge separation. The charge transfer of Cu<sub>x</sub>O-TiO<sub>2</sub> heterojunction is shown in Fig. 13b. Here, e<sup>-</sup> could transfer from the CB of Cu<sub>x</sub>O to TiO<sub>2</sub> and react with CO<sub>2</sub> to produce CH<sub>4</sub>. Schaak

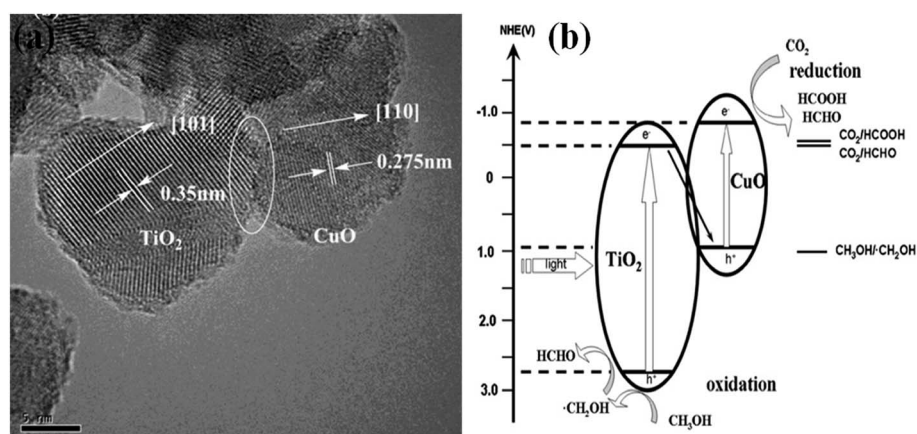


Fig. 12 (a) HRTEM image of CuO/TiO<sub>2</sub> NTs; (b) schematic of the charge separation of CuO/TiO<sub>2</sub> NT heterojunction. Reproduced from ref. 157 with permission from Elsevier.

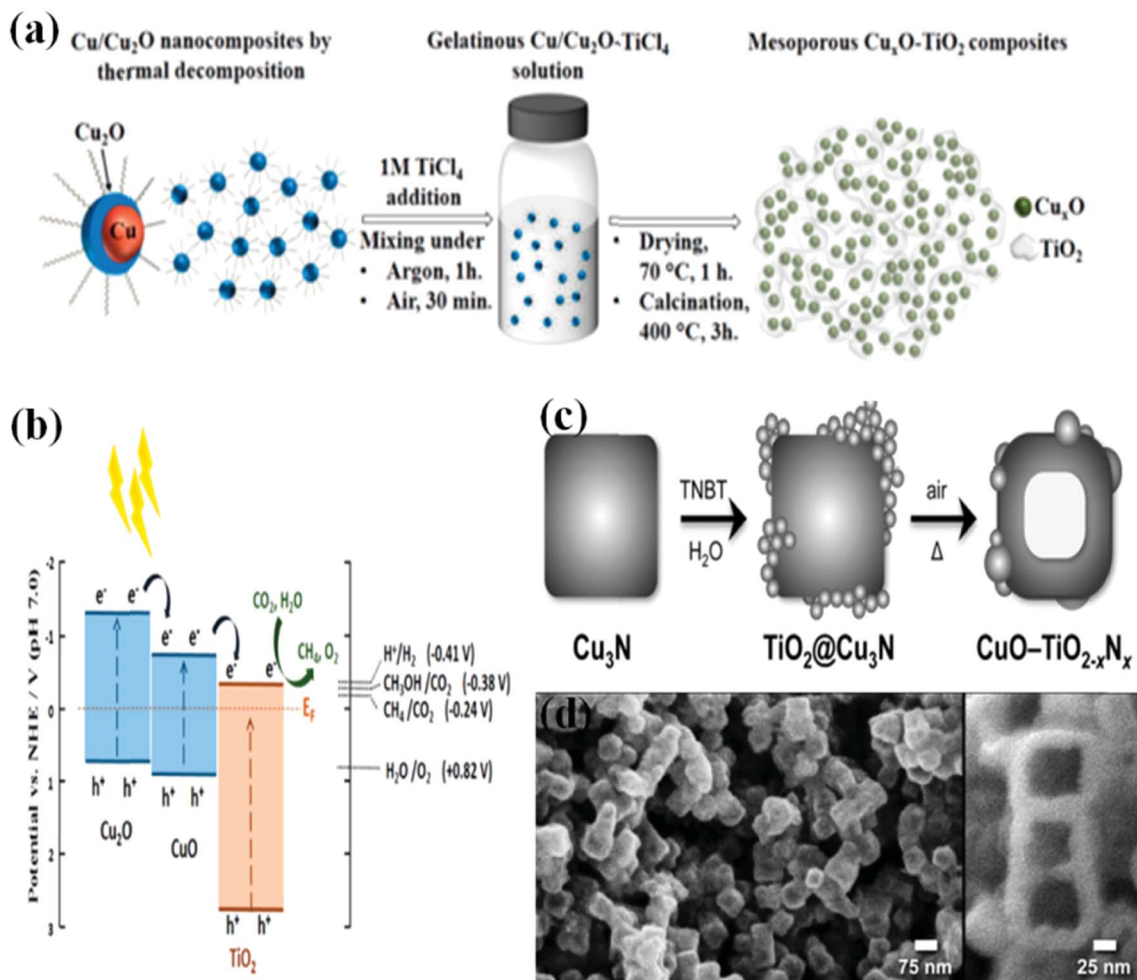


Fig. 13 (a) Schematic view of the synthesis of mesoporous  $\text{Cu}_x\text{O}-\text{TiO}_2$  photocatalysts; (b) schematic of the charge transfer in the  $\text{Cu}_x\text{O}-\text{TiO}_2$  heterojunction; (c) schematic view of the synthesis of  $\text{CuO}-\text{TiO}_{2-x}\text{N}_x$  hollow nanocubes; (d) FESEM images of the  $\text{CuO}-\text{TiO}_{2-x}\text{N}_x$  hollow nanocubes. (a and b) Reproduced from ref. 160 with permission from the American Chemical Society. (c and d) Reproduced from ref. 161 with permission from Wiley-VCH.

*et al.* reported novel hollow  $\text{CuO}$  nanocubes loaded over titanium oxynitride ( $\text{TiO}_{2-x}\text{N}_x$ ) by a multi-template strategy (Fig. 13c).<sup>161</sup> Fig. 13d shows the FESEM images of the  $\text{CuO}-\text{TiO}_{2-x}\text{N}_x$  hollow nanocubes. N doped onto  $\text{TiO}_2$  could extend the light absorption capability into the visible light region. The photocurrent density of a thin film of  $\text{CuO}-\text{TiO}_{2-x}\text{N}_x$  under AM 1.5G irradiation could confirm the formed p-n heterojunction by electrochemical evidence. The  $\text{CuO}-\text{TiO}_{2-x}\text{N}_x$  hollow nanocubes exhibited a higher  $\text{CH}_4$  production rate ( $41.3 \text{ ppm g}^{-1} \text{ h}^{-1}$ ) than Degussa P25 ( $16.2 \text{ ppm g}^{-1} \text{ h}^{-1}$ ).

In a previous work, GaP was used for the photoelectrochemical reduction of  $\text{CO}_2$  in 1978. However, it is rarely used as a photocatalyst owing to the low oxidizing power of its VB.<sup>162</sup> Here,  $e^-$  in the CB can convert  $\text{CO}_2$  into  $\text{CH}_4$ , but  $h^+$  in the VB is unable to oxidize  $\text{H}_2\text{O}$ ; therefore, bare GaP cannot induce the photocatalytic reduction of  $\text{CO}_2$ . GaP coupled with  $\text{TiO}_2$  that forms a p-n heterojunction for the photocatalytic reduction of  $\text{CO}_2$  was fabricated by Marcì *et al.* for the first time.<sup>163</sup> GaP/ $\text{TiO}_2$  composites could effectively exhibit both the reduction of  $\text{CO}_2$

and oxidation of  $\text{H}_2\text{O}$ , and the p-n heterojunction could promote the separation of  $e^-/h^+$  pairs. The highest photocatalytic performance was obtained with 10 wt% GaP, which was ascribed to the band structures and charge transfer between GaP and  $\text{TiO}_2$ .

**4.1.2 Non-p-n heterojunction.** Non-p-n heterojunction is another type of heterojunction. Owing to the staggered alignment of the energy levels, non-p-n heterojunction structures have been already used for promoting the performance of the photocatalytic reduction of  $\text{CO}_2$  into solar fuels.

In previous studies, sulfide semiconductors have received considerable attention. Furthermore, many sulfides (*e.g.*, CdS, PbS) have a narrow bandgap, where the onset of absorption begins in the visible light or even infrared region. However, heavy metal ions, *e.g.*, Pb(II), Hg(II), Cr(VI), and Cd(II), exhibit obvious non-biodegradation and bioaccumulation and cannot easily degrade into cleaner products, which cause severe water pollution and induce serious threats to living organisms.<sup>164-167</sup> Moreover, bare sulfide semiconductors exhibit photocorrosion.

Recent studies have confirmed that sulfide semiconductors are promising materials for the photosensitization of TiO<sub>2</sub> in the photocatalytic reduction of CO<sub>2</sub>.<sup>168</sup> Coupling with TiO<sub>2</sub> to form a heterojunction can both overcome the photocorrosion of sulfide semiconductors and improve the photocatalytic activity of TiO<sub>2</sub>. Yin *et al.* synthesized the CdS/TiO<sub>2</sub> heterojunction *via* two-step hydrothermal methods, and its synthesis process is shown in Fig. 14a.<sup>169</sup> Fig. 14b and c show the TEM images of TiO<sub>2</sub> nanosheets and CdS/TiO<sub>2</sub> heterojunction composites. The sizes of the TiO<sub>2</sub> nanosheet were about 50–60 nm. The HRTEM image of the CdS/TiO<sub>2</sub> heterojunction composite is shown in Fig. 14d, where the typical crystallized cubic CdS {002} and anatase TiO<sub>2</sub> {010} planes with the characteristic lattice spacing of 0.334 nm and 0.378 nm, respectively, were observed.<sup>170</sup> The CdS NPs were dispersed on the surface and edge of the TiO<sub>2</sub> nanosheet, and the interface could form a non-p–n heterojunction. Cyclohexanol was used as the solvent and reductant because the solubility of CO<sub>2</sub> in cyclohexanol was much higher than that in water at room temperature. With the optimum molar ratio of TiO<sub>2</sub> to CdS (8 : 1), it exhibited the maximum cyclohexyl formate (CF) yield. The band energy values and charge carrier transfer of CdS/TiO<sub>2</sub> are shown in Fig. 14e. The high separation and transfer of e<sup>-</sup>/h<sup>+</sup> pairs in this heterojunction were crucial for its high photocatalytic activity.

A similar finding about CdS/TiO<sub>2</sub> for the photocatalytic reduction of CO<sub>2</sub> was reported by Fatemi *et al.*<sup>171</sup> Here, 45% CdS/TiO<sub>2</sub> exhibited the maximum CO and CH<sub>4</sub> formation rates under both UV-vis and visible light irradiations. In another work, Li *et al.* reported the CdS or Bi<sub>2</sub>S<sub>3</sub> loaded onto TiO<sub>2</sub> NTs

*via* a direct precipitation method.<sup>172</sup> The CH<sub>3</sub>OH yields of CdS/TiO<sub>2</sub> and Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> were about 159.5 and 224.6 μmol L<sup>-1</sup>, respectively, which were higher than those of P25 (91 μmol L<sup>-1</sup>) and TiO<sub>2</sub> NTs (102.5 μmol L<sup>-1</sup>). The loaded CdS or Bi<sub>2</sub>S<sub>3</sub> could improve the visible light absorption and photocatalytic activity. The different sizes of PbS quantum dots (QDs) loaded onto TiO<sub>2</sub> for the photocatalytic reduction of CO<sub>2</sub> was reported by Wang *et al.*<sup>173</sup> It was interesting to note that 4 nm PbS-QD-sensitized TiO<sub>2</sub> yielded a higher CO<sub>2</sub> conversion rate than the 3 and 5 nm counterparts because of the synergetic effects of good light absorption and high charge separation properties. In particular, the decreased activity with larger particles was attributable to the smaller surface areas and longer electron migration paths, whereas the decreased activity with smaller particles was associated with changes in the optical and electronic properties.

Zhao *et al.* synthesized CeO<sub>2</sub> loaded onto 3D ordered macroporous (3DOM) TiO<sub>2</sub> *via* the gas bubbling-assisted membrane precipitation (GBMP) method.<sup>174</sup> The SEM images of 3DOM TiO<sub>2</sub> and CeO<sub>2</sub>/TiO<sub>2</sub> are shown in Fig. 15a and b, respectively. From Fig. 15a, it is evident that 3DOM TiO<sub>2</sub> had periodic voids, where the average diameter and wall thickness were about 310 ± 10 nm and 40 ± 5 nm, respectively. After the loading of CeO<sub>2</sub>, the morphology did not obviously change. The result of UV-vis DRS indicated that 3DOM CeO<sub>2</sub>/TiO<sub>2</sub> had a good photoresponse from the UV to the visible light region. 3DOM CeO<sub>2</sub>/TiO<sub>2</sub> possessed high photocatalytic activity. The mechanism for the enhanced activity of 3DOM CeO<sub>2</sub>/TiO<sub>2</sub> is shown in Fig. 15c. In a similar study by Wang *et al.*, the mesoporous CeO<sub>2</sub>/TiO<sub>2</sub> photocatalyst with a 2D hexagonal structure was synthesized *via*

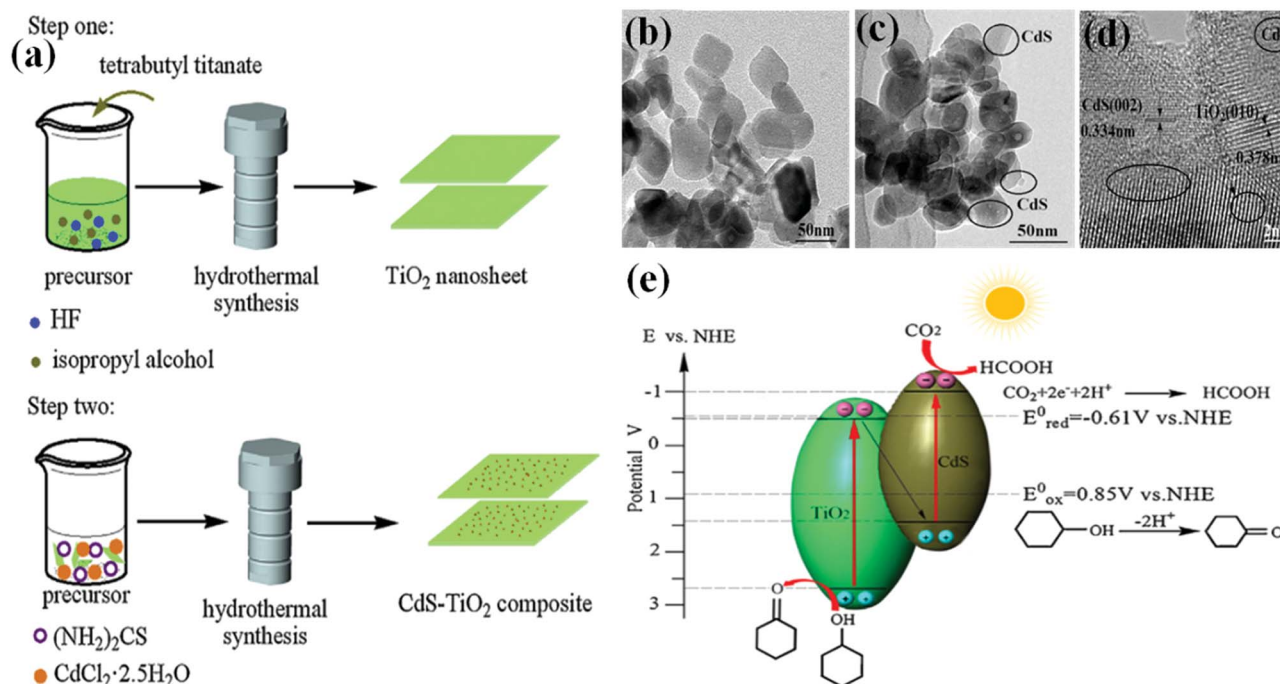


Fig. 14 (a) Schematic view of the synthesis of CdS–TiO<sub>2</sub> composites; TEM images of (b) TiO<sub>2</sub> nanosheets and (c) CdS–TiO<sub>2</sub> composites; (d) HRTEM image of CdS–TiO<sub>2</sub> composite; (e) band energy values and charge carrier transfer of CdS/TiO<sub>2</sub> composites. Reproduced from ref. 169 with permission from Elsevier.

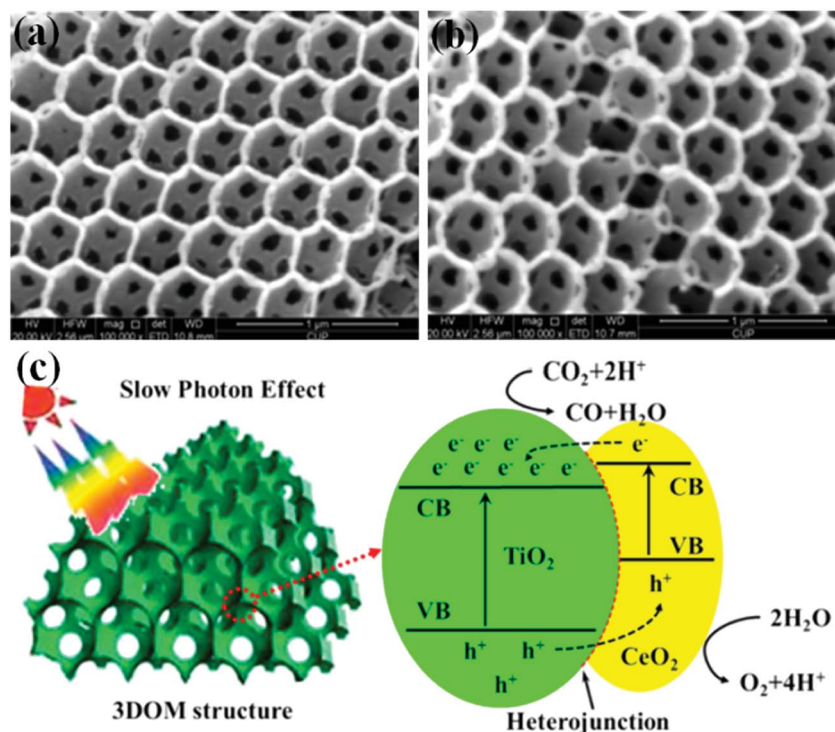


Fig. 15 SEM images of (a) 3DOM TiO<sub>2</sub> and (b) CeO<sub>2</sub>-TiO<sub>2</sub>; (c) mechanism for the enhanced activity of 3DOM CeO<sub>2</sub>/TiO<sub>2</sub>. Reproduced from ref. 174 with permission from the American Chemical Society.

a nanocasting method.<sup>175</sup> Its photocatalytic activity was much higher than that of commercial P25.

When compared with sulfides, many oxides do not undergo photooxidation under light irradiation and have been widely used in photocatalytic reduction and oxidation processes. For example, zinc oxide (ZnO) is an n-type semiconductor with a direct bandgap ( $\sim 3.3$  eV), and it is usually used in photocatalytic oxidation processes.<sup>176</sup> ZnO hybrid photocatalysts have aroused considerable attention due to its high effective separation of  $e^-/h^+$  pairs. Ye *et al.* synthesized TiO<sub>2</sub>/ZnO mesoporous “French fries” (MFFs) hybrid photocatalyst *via* a facile furfural alcohol-derived polymerization-oxidation (FAPO) method.<sup>177</sup> The CH<sub>4</sub> formation rate over TiO<sub>2</sub>/ZnO MFFs was  $55 \mu\text{mol g}^{-1} \text{h}^{-1}$ , which was 5.9 times higher than that of commercial P25 ( $9.3 \mu\text{mol g}^{-1} \text{h}^{-1}$ ). In another work, Wu *et al.* reported V and W doped onto TiO<sub>2</sub> and then coated onto a honeycomb support.<sup>178</sup> After doping with V and W, the visible light absorption and charge separation improved. V<sub>2</sub>O<sub>5</sub> and TiO<sub>2</sub> could form a heterojunction and doping W<sup>6+</sup> could trap  $e^-$  and react with CO<sub>2</sub> and H<sub>2</sub>O to yield solar fuels. Therefore, the photocatalytic activity was improved.

FeTiO<sub>3</sub>/TiO<sub>2</sub> heterojunction photocatalysts were synthesized by Ling *et al.*<sup>179</sup> The highest CH<sub>3</sub>OH yield of 10% FeTiO<sub>3</sub>/TiO<sub>2</sub> was  $0.462 \mu\text{mol g}^{-1} \text{h}^{-1}$  and  $0.432 \mu\text{mol g}^{-1} \text{h}^{-1}$  under UV-vis and visible light irradiations, respectively. Due to the unique band structure and heterojunction over FeTiO<sub>3</sub>/TiO<sub>2</sub>, the CH<sub>3</sub>OH yield was obviously improved. Ye *et al.* reported SrTiO<sub>3</sub> (STO) loaded onto TiO<sub>2</sub> NT arrays (TNA) *via* a facile hydrothermal method and then Au–Cu bimetallic alloys were loaded

on the STO/TNA *via* the microwave-assisted solvothermal route.<sup>180</sup> The Au<sub>3</sub>Cu@STO/TNA exhibited good photocatalytic activity for the generation of CO and CH<sub>4</sub> under UV-vis light irradiation. Owing to the alloying effect, the Au–Cu alloy yielded higher activity than that of pure Au and Cu. STO and TiO<sub>2</sub> could form a heterojunction to facilitate photogenerated carrier separation.

Graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) with a moderate bandgap energy ( $\sim 2.7$ – $2.8$  eV) has aroused considerable attention for photocatalytic applications in the last couple of years. However, its photocatalytic activity is very low because of the high recombination of charge carriers. Various methods, such as using a heterojunction, have been investigated to enhance the photocatalytic activity of g-C<sub>3</sub>N<sub>4</sub>. Reli *et al.* synthesized TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> *via* the simple mechanical mixing of g-C<sub>3</sub>N<sub>4</sub> with TiO<sub>2</sub>.<sup>181</sup> Further, (0.3/1) TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> showed good photocatalytic activity for the photocatalytic reduction of CO<sub>2</sub> to produce CH<sub>4</sub> and CO. In a similar work, Zhao *et al.* tuned the mass ratios of urea and Ti(SO<sub>4</sub>)<sub>2</sub> to synthesize g-C<sub>3</sub>N<sub>4</sub>-N-TiO<sub>2</sub> heterojunction catalysts.<sup>182</sup> It was interesting to find that CO was the main product of the g-C<sub>3</sub>N<sub>4</sub>-N-TiO<sub>2</sub> heterojunction, whereas CH<sub>4</sub> and CO were the main products of N-TiO<sub>2</sub>. Therefore, the g-C<sub>3</sub>N<sub>4</sub> could suppress CH<sub>4</sub> formation.

Recently, colorful inorganic materials with high stability and strong absorption in the visible light region have been used for modifying TiO<sub>2</sub>. AgBr often serves as a photosensitive material to improve the activity. He *et al.* reported the AgBr/TiO<sub>2</sub> photocatalyst for CO<sub>2</sub> reduction to CH<sub>4</sub>, CH<sub>3</sub>OH, CH<sub>3</sub>CH<sub>2</sub>OH and CO under visible light irradiation.<sup>183</sup> It was interesting to find

that 23.2% AgBr/TiO<sub>2</sub> exhibited the maximum yield and high stability.

**4.1.3 Z-scheme heterojunction.** Lately, another Z-scheme heterojunction system has been developed for the photocatalytic reduction of CO<sub>2</sub>.<sup>184,185</sup> Gondal *et al.* first synthesized a direct Z-scheme Si/TiO<sub>2</sub> *via* a hydrothermal method.<sup>186</sup> The direct Z-scheme Si/TiO<sub>2</sub> exhibited better photocatalytic performance for the photocatalytic reduction of CO<sub>2</sub> into CH<sub>3</sub>OH than that of individual Si and TiO<sub>2</sub>, because this Z-scheme system could considerably enhance the separation efficiency of e<sup>-</sup>/h<sup>+</sup> pairs and prolong their longevity. In another report, Xin *et al.* fabricated ZnFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> nanobelts Z-scheme heterojunction photocatalyst *via* two-step hydrothermal methods.<sup>187</sup> The obtained ZnFe<sub>2</sub>O<sub>4</sub>/TiO<sub>2</sub> exhibited high photocatalytic performance in cyclohexanol for the photocatalytic reduction of CO<sub>2</sub> under UV light irradiation. Furthermore, the introduction of ZnFe<sub>2</sub>O<sub>4</sub> could extend the response of TiO<sub>2</sub> into the visible light region.

The all-solid-state Z-scheme CdS/reduced graphene oxide (rGO)/TiO<sub>2</sub> core-shell nanostructure for the photocatalytic reduction of CO<sub>2</sub> into CH<sub>4</sub> was reported by Zou *et al.*, where rGO served as the electron mediator to accelerate the separation of e<sup>-</sup>/h<sup>+</sup> pairs and improve the stability of CdS.<sup>188</sup> The charge transfer in CdS/rGO/TiO<sub>2</sub> Z-scheme heterojunction system is shown in Fig. 16. Under UV-vis light irradiation, both CdS and TiO<sub>2</sub> could produce e<sup>-</sup> and h<sup>+</sup>. As the CB of TiO<sub>2</sub> was negative than the Fermi level of rGO, e<sup>-</sup> could transfer from the CB of TiO<sub>2</sub> to rGO and then recombine with h<sup>+</sup> in the VB of CdS. This process effectively improved h<sup>+</sup> in TiO<sub>2</sub> for the oxidation of H<sub>2</sub>O and e<sup>-</sup> in CdS for the photocatalytic reduction of CO<sub>2</sub>.

## 4.2 TiO<sub>2</sub>-metal heterojunction

Metals and semiconductors have different physical and chemical properties. When they couple with each other, a heterojunction (Schottky barrier) is produced, which can produce a space-charge separation region.<sup>136</sup> The Schottky barrier also works as an electron trap to reduce the recombination of e<sup>-</sup>/h<sup>+</sup> pairs. Further, n-type semiconductors, such as TiO<sub>2</sub>, usually form this heterojunction. Moreover, the work function of

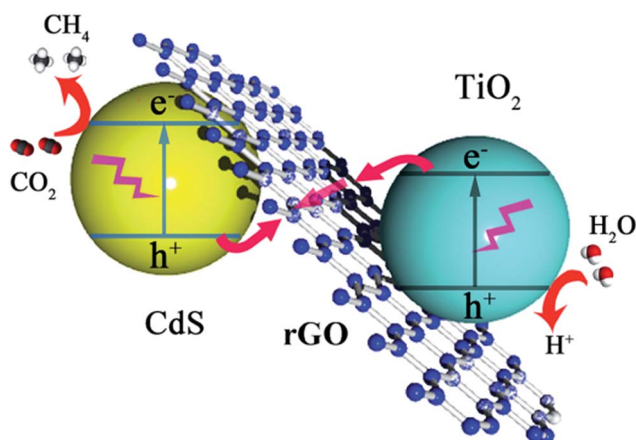


Fig. 16 Schematic of CdS/rGO/TiO<sub>2</sub> Z-scheme system. Reproduced from ref. 188 with permission from The Royal Society of Chemistry.

a metal is often higher than that of TiO<sub>2</sub>. A metal can also produce e<sup>-</sup> by the localized surface plasmon resonance (SPR) excitation under light irradiation. The schematic of TiO<sub>2</sub>-metal heterojunction is shown in Fig. 17.<sup>136</sup> Here, e<sup>-</sup> can flow from TiO<sub>2</sub> to the metal in order to align the Fermi energy levels. Therefore, metal and TiO<sub>2</sub> are excessively negative and positive, respectively. Further, e<sup>-</sup> can aggregate in the CB of TiO<sub>2</sub> and react with CO<sub>2</sub> and H<sup>+</sup> to produce solar fuels; h<sup>+</sup> in the VB of TiO<sub>2</sub> reacts with H<sub>2</sub>O to generate H<sup>+</sup> and O<sub>2</sub>. Table 4 summarizes and compares the typical TiO<sub>2</sub>-metal heterojunction photocatalysts for the photocatalytic reduction of CO<sub>2</sub>.

The activation of CO<sub>2</sub> is more efficiently performed on TiO<sub>2</sub> decorated with noble-metal NPs. Zhao *et al.* prepared uniform Au NPs supported on 3DOM TiO<sub>2</sub> *via* a facile GBMR method.<sup>189</sup> Fig. 18A shows the UV-vis DRS of P25, 3DOM TiO<sub>2</sub>, and 3DOM Au/TiO<sub>2</sub>. It was clearly observed that 3DOM TiO<sub>2</sub> exhibited a slight red-shift in comparison with P25, and 3DOM Au/TiO<sub>2</sub> had a broad band centered by the Au SPR effect.<sup>190</sup> It was interesting to find that increasing the Au amounts could remarkably improve the visible light absorbance. Fig. 18B shows the CH<sub>4</sub> production amount under visible light irradiation. The 3DOM Au<sub>8</sub>/TiO<sub>2</sub> exhibited the highest CH<sub>4</sub> formation rate (2.89 μmol g<sup>-1</sup> h<sup>-1</sup>), which was 2.61 and 2.03 times higher than those of P25 (1.11 μmol g<sup>-1</sup> h<sup>-1</sup>) and 3DOM TiO<sub>2</sub> (1.42 μmol g<sup>-1</sup> h<sup>-1</sup>), respectively. The photocatalytic mechanism of 3DOM Au/TiO<sub>2</sub> is shown in Fig. 18C. The 3DOM TiO<sub>2</sub> structure could increase the light absorption and diffusion of CO<sub>2</sub>, and Au also extended the spectral range from the UV to the visible light region.

Charge recombination is an important negative effect on the photocatalytic performance. Chai *et al.* synthesized Ag-doped multiwalled carbon nanotubes (MWCNT)@TiO<sub>2</sub> core-shell photocatalysts *via* a one-step method with the core-shell coating approach, where MWCNT and TiO<sub>2</sub> served as the core and shell, respectively.<sup>191</sup> The HRTEM image of Ag-doped MWCNT@TiO<sub>2</sub> (Fig. 19a) showed the Ag-TiO<sub>2</sub> heterojunction. In Fig. 19b, the CH<sub>4</sub> formation rate was evaluated with respect to the photocatalytic activity. It was revealed that the Ag-doped MWCNT@TiO<sub>2</sub> had good stability and 2 wt% Ag doping exhibited the highest CH<sub>4</sub> formation rate under visible light irradiation. Fig. 19c shows the charge transfer route in Ag-MWCNT@TiO<sub>2</sub>. Further, e<sup>-</sup> could transfer to the MWCNT and

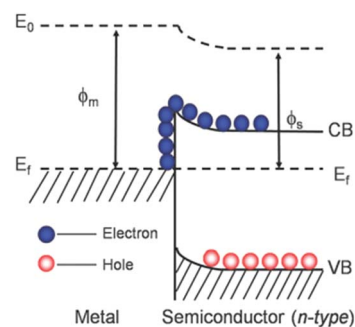


Fig. 17 Schematic of a Schottky barrier. Reproduced from ref. 136 with permission from The Royal Society of Chemistry.

Table 4 Comparison of typical TiO<sub>2</sub>-metal heterojunction systems

| Photocatalyst               | Solvent          | Light source          | Time <sup>a</sup> | Formation rate <sup>b</sup>                                   | QE <sup>c</sup> (%) | Reference |
|-----------------------------|------------------|-----------------------|-------------------|---------------------------------------------------------------|---------------------|-----------|
| 3DOM Au/TiO <sub>2</sub>    | H <sub>2</sub> O | Xe lamp               | 8                 | CH <sub>4</sub> : 2.89                                        | —                   | 189       |
| Ag-MWCNT@TiO <sub>2</sub>   | H <sub>2</sub> O | Energy saving light   | 7.5               | CH <sub>4</sub> : 0.91; C <sub>2</sub> H <sub>4</sub> : 0.048 | —                   | 191       |
| CdSe/Pt/TiO <sub>2</sub>    | H <sub>2</sub> O | Xe lamp (λ > 420 nm)  | 6                 | CH <sub>4</sub> : 48 ppm g <sup>-1</sup> h <sup>-1</sup>      | —                   | 193       |
| RGO/Pt-TiO <sub>2</sub> NTs | H <sub>2</sub> O | Tungsten halogen lamp | 5                 | CH <sub>4</sub> : 2.88 μmol m <sup>-2</sup> h <sup>-1</sup>   | —                   | 194       |

<sup>a</sup> Irradiation time, h. <sup>b</sup> Formation rate, μmol g<sup>-1</sup> h<sup>-1</sup>. <sup>c</sup> Quantum efficiency (%).

Ag to reduce the recombination rate of e<sup>-</sup>/h<sup>+</sup> pairs, improving the photocatalytic performance.

It has been reported that semiconductor nanocrystal QDs that can sensitize TiO<sub>2</sub> can make full use of the visible light spectrum and increase charge separation efficiency.<sup>192</sup> Wang *et al.* synthesized Pt/TiO<sub>2</sub> via a wet impregnation method and then CdSe QDs were assembled on Pt/TiO<sub>2</sub>.<sup>193</sup> Under visible light irradiation, CdSe/Pt/TiO<sub>2</sub> exhibited a high CH<sub>4</sub> formation rate. In another study, Saravanan *et al.* reported rGO loaded onto Pt/TiO<sub>2</sub> NTs for CO<sub>2</sub> reduction under visible light irradiation. Pt NPs could increase the visible light absorption by the LSPR effect and rGO could be used as an electron reservoir to reduce the recombination rate of e<sup>-</sup>/h<sup>+</sup> pairs.<sup>194</sup>

### 4.3 TiO<sub>2</sub>-carbon heterojunction

GR is a new class of material in carbon atoms, which has a two-dimensional sp<sup>2</sup>-hybridized carbon atom in the carbon family.<sup>195-198</sup> It has some unique properties, such as high surface area (2600 m<sup>2</sup> g<sup>-1</sup>),<sup>199-202</sup> high thermal conductivity (5000 W m<sup>-1</sup> K<sup>-1</sup>),<sup>203,204</sup> high mobility of charge carriers (200 000 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>),<sup>205,206</sup> and good optical transparency (97.7%).<sup>207</sup> It was first prepared by Novoselov *et al.*,<sup>208</sup> it has received increased attention due to its potential applications. Furthermore, GR can also improve the separation of e<sup>-</sup>/h<sup>+</sup> pairs and suppress the recombination of e<sup>-</sup>/h<sup>+</sup> pairs. Table 5 summarizes and compares the typical TiO<sub>2</sub>-carbon heterojunction photocatalysts for the photocatalytic reduction of CO<sub>2</sub>.

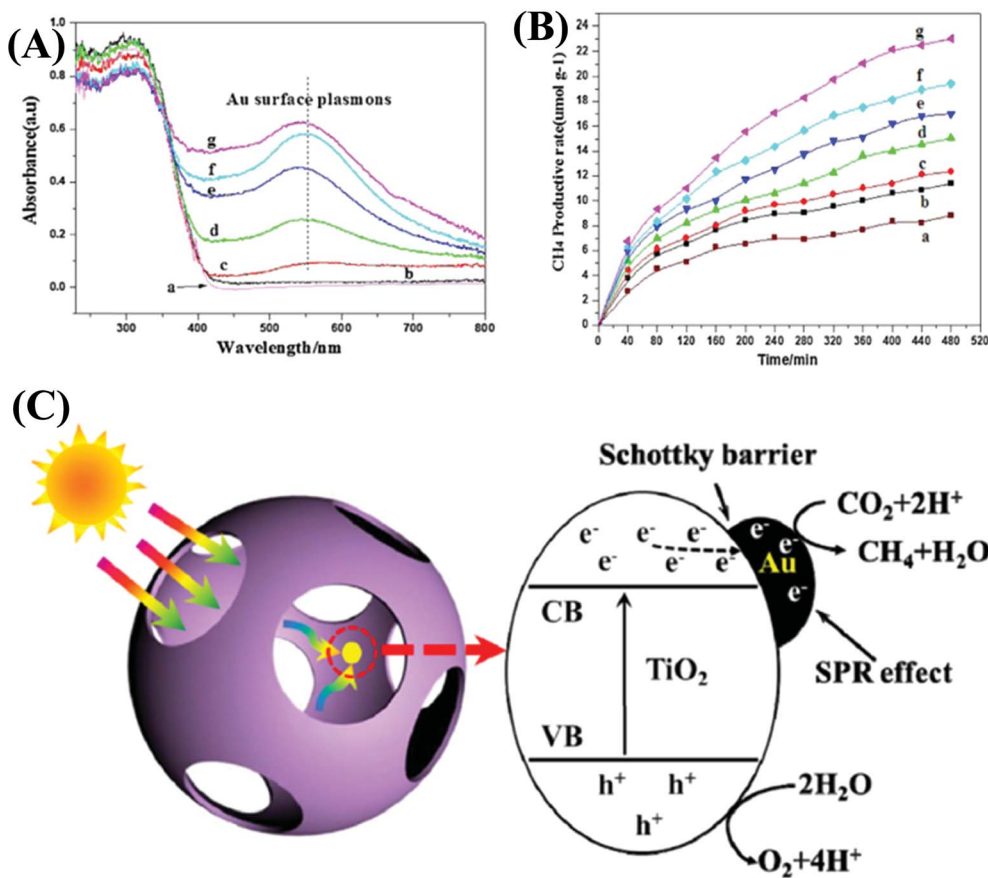


Fig. 18 (A) UV-vis DRS of 3DOM Au/TiO<sub>2</sub>; (B) CH<sub>4</sub> production amounts of 3DOM Au/TiO<sub>2</sub>; (C) possible photocatalytic mechanism of 3DOM Au/TiO<sub>2</sub>. (a) P25; (b) TiO<sub>2</sub>; (c) Au<sub>0.5</sub>/TiO<sub>2</sub>; (d) Au<sub>1</sub>/TiO<sub>2</sub>; (e) Au<sub>2</sub>/TiO<sub>2</sub>; (f) Au<sub>4</sub>/TiO<sub>2</sub>; (g) Au<sub>8</sub>/TiO<sub>2</sub>. Reproduced from ref. 189 with permission from Elsevier.



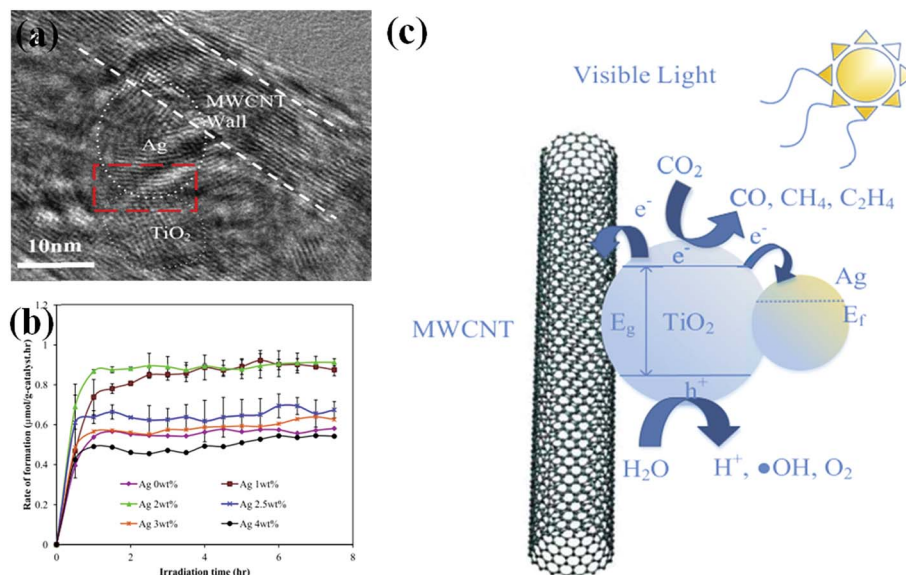


Fig. 19 (a) HRTEM image of 2 wt% Ag-MWCNT@TiO<sub>2</sub>; (b) CH<sub>4</sub> formation rate over Ag-MWCNT@TiO<sub>2</sub>; (c) schematic of charge transfer in Ag-MWCNT@TiO<sub>2</sub>. Reproduced from ref. 191 with permission from Elsevier.

The Cu<sub>2</sub>O/GR/TNA heterostructure for the photocatalytic reduction of CO<sub>2</sub> under visible light irradiation was reported by Cao *et al.*<sup>209</sup> The fabrication steps for the Cu<sub>2</sub>O/GR/TNA heterostructure is shown in Fig. 20a. TNA was first synthesized by two-step anodization methods on Ti foils and then GR was loaded by cyclic voltammetric reduction. Lastly, Cu<sub>2</sub>O was loaded by an electrochemical deposition method. It was found that the highest CH<sub>3</sub>OH yield was achieved by the Cu<sub>2</sub>O/GR/TNA heterostructure (Fig. 20b). Under visible light irradiation, e<sup>-</sup> and h<sup>+</sup> were produced by Cu<sub>2</sub>O. With the help of GR, e<sup>-</sup> could transfer to the CB of TNA; therefore, e<sup>-</sup> and h<sup>+</sup> were effectively separated. The possible mechanism of Cu<sub>2</sub>O/GR/TNA is proposed in Fig. 20c. The ternary heterojunction could increase the visible light absorption and suppress the recombination of e<sup>-</sup>/h<sup>+</sup> pairs. In a similar study, Chai *et al.* reported N-doped TiO<sub>2</sub> with exposed {001} facet (N-TiO<sub>2</sub>-001) loaded onto GR sheets to fabricate the N-TiO<sub>2</sub>-001/GR catalyst *via* the solvothermal method.<sup>210</sup> The N-TiO<sub>2</sub>-001/GR showed the highest photocatalytic activity due to the high catalytic activity of {001} facet and GR.

Graphene oxide (GO) is a ramification of GR, which has abundant oxygenated functional groups (*e.g.*, -OH, -COOH) on

its surface and large surface area.<sup>211–213</sup> Moreover, we often use GO as the precursor to fabricate GR–semiconductor composite photocatalysts.<sup>214–216</sup> Under chemical or thermal conditions, GO can be readily reduced to GR, which is commonly referred to as rGO.<sup>217</sup> The rGO–TiO<sub>2</sub> hybrid photocatalyst was synthesized by Chai *et al.* *via* a solvothermal method.<sup>218</sup> rGO–TiO<sub>2</sub> had a high CH<sub>4</sub> formation rate (0.135 μmol g<sup>-1</sup> h<sup>-1</sup>) under visible light irradiation. The intimate interface between rGO and TiO<sub>2</sub> largely improved the transfer of e<sup>-</sup> from TiO<sub>2</sub> to rGO, resulting in the enhanced CH<sub>4</sub> formation rate. In a further study, Chai *et al.* synthesized GO-doped O-rich TiO<sub>2</sub> (GO–OTiO<sub>2</sub>) heterojunction structure *via* a chemical impregnation method.<sup>219</sup> It was found that 5 wt% GO–OTiO<sub>2</sub> yielded the highest CH<sub>4</sub> formation rate (0.29 μmol g<sup>-1</sup> h<sup>-1</sup>) and good photostability. In another report, noble metals (Pt, Pd, Ag, and Au) loaded onto rGO/TiO<sub>2</sub> (GT) were also synthesized by Chai *et al.*, where Pt achieved the highest CH<sub>4</sub> yield among these noble metals.<sup>220</sup> The photocatalytic performance was strongly related to the work function of the metal, where Pt had the highest work function and e<sup>-</sup> could effectively transfer from TiO<sub>2</sub> to Pt. The possible mechanism of Pt–GT for the photocatalytic reduction of CO<sub>2</sub> to CH<sub>4</sub> is shown in Fig. 21. Under light irradiation, TiO<sub>2</sub> could

Table 5 Comparison of typical TiO<sub>2</sub>–carbon heterojunction systems

| Photocatalyst                                   | Solvent                | Light source       | Time <sup>a</sup> | Formation rate <sup>b</sup>                                  | QE <sup>c</sup> (%) | Reference |
|-------------------------------------------------|------------------------|--------------------|-------------------|--------------------------------------------------------------|---------------------|-----------|
| Cu <sub>2</sub> O/graphene/TiO <sub>2</sub> NTs | 1 M NaHCO <sub>3</sub> | Xe lamp            | 6                 | CH <sub>3</sub> OH: 45 μmol cm <sup>-2</sup> h <sup>-1</sup> | —                   | 209       |
| N-TiO <sub>2</sub> -001/GR                      | H <sub>2</sub> O       | Energy-saving lamp | 10                | CH <sub>4</sub> : 0.37                                       | —                   | 210       |
| rGO–TiO <sub>2</sub>                            | H <sub>2</sub> O       | Energy-saving bulb | 5                 | CH <sub>4</sub> : 0.135                                      | —                   | 218       |
| GO–OTiO <sub>2</sub>                            | H <sub>2</sub> O       | Energy-saving bulb | 6                 | CH <sub>4</sub> : 0.29                                       | —                   | 219       |
| Pt–rGO–TiO <sub>2</sub>                         | H <sub>2</sub> O       | Energy-saving bulb | 6                 | CH <sub>4</sub> : 0.28                                       | —                   | 220       |
| N-CQDs–TiO <sub>2</sub>                         | H <sub>2</sub> O       | Xe lamp            | 6                 | CH <sub>4</sub> : 3.98; CO: 6.13                             | —                   | 223       |

<sup>a</sup> Irradiation time, h. <sup>b</sup> Formation rate, μmol g<sup>-1</sup> h<sup>-1</sup>. <sup>c</sup> Quantum efficiency (%).

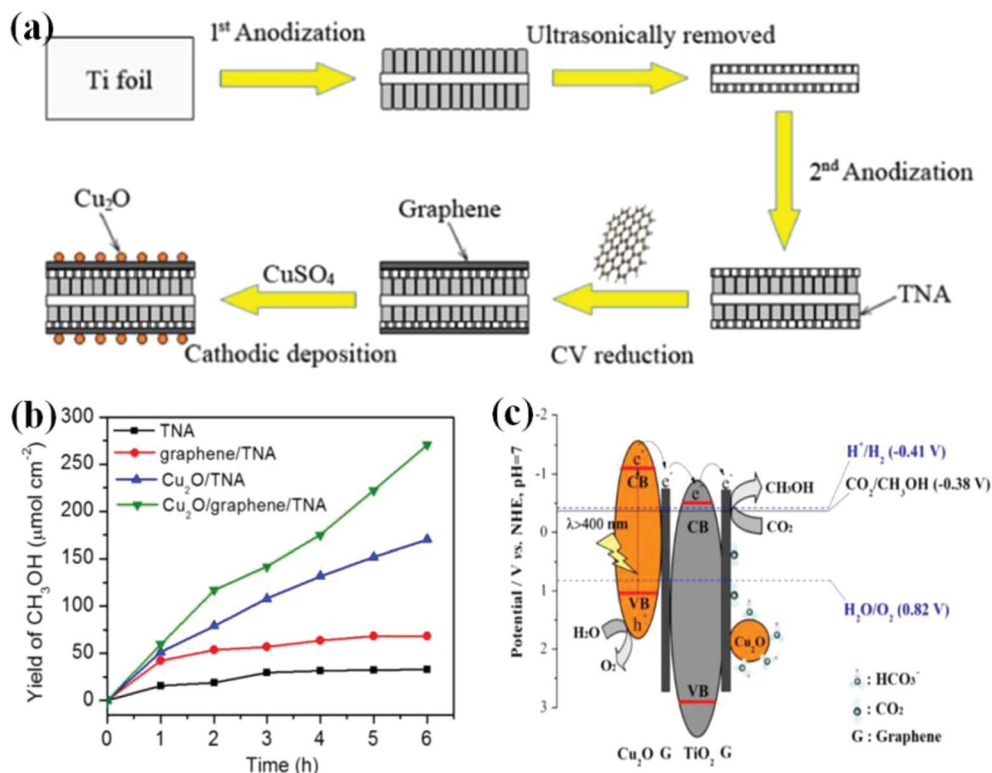


Fig. 20 (a) Fabrication steps for Cu<sub>2</sub>O/GR/TNA heterostructure; (b) photocatalytic CH<sub>3</sub>OH evolution over 6 h for Cu<sub>2</sub>O/GR/TNA heterostructure; (c) possible mechanism of Cu<sub>2</sub>O/GR/TNA. Reproduced from ref. 209 with permission from Elsevier.

produce e<sup>-</sup> and h<sup>+</sup>. Moreover, Pt could effectively trap e<sup>-</sup> to react with CO<sub>2</sub> and H<sup>+</sup> to produce CH<sub>4</sub> (path 1). Furthermore, Pt was also deposited onto rGO sheets. Therefore, e<sup>-</sup> could pass the two-dimensional carbon network to rGO sheets and then get trapped by the Pt sites to react with CO<sub>2</sub> and H<sup>+</sup> (path 2).

Carbon quantum dots (CQDs) are a new class of material in the carbon family, which exhibit good efficiency of charge separation and electron reservoir properties.<sup>221</sup> These CQDs can absorb visible and near-infrared (NIR) light in the solar spectrum.<sup>222</sup> Therefore, coupling CQDs with TiO<sub>2</sub> can ensure full use of the solar spectrum. Cao *et al.* reported N-CQDs/TiO<sub>2</sub> heterojunctions for the photocatalytic reduction of CO<sub>2</sub>,<sup>223</sup> CH<sub>4</sub> and CO were the major products under simulated solar light irradiation. These N-CQDs could serve as the photosensitizer and electrons reservoir to enhance the photocatalytic activity.

#### 4.4 Phase or facet TiO<sub>2</sub> heterojunction

**4.4.1 Phase heterojunction.** The phase heterojunction structure only exists in one semiconductor with two or more different crystal phases. Recently, the phase heterojunction has attracted considerable attention for the photocatalytic reduction of CO<sub>2</sub>.<sup>224–226</sup> TiO<sub>2</sub> has the anatase, brookite, and rutile phases. Anatase-based catalysts usually have high activity when compared with rutile-based catalysts owing to the fast recombination of e<sup>-</sup>/h<sup>+</sup> pairs in the rutile phase. Further, brookite-based catalysts are rarely studied because it is very difficult to synthesize high-purity brookite.<sup>227</sup> Table 6 summarizes and

compares the typical phase or facet of TiO<sub>2</sub> heterojunction photocatalysts for the photocatalytic reduction of CO<sub>2</sub>.

Bai *et al.* reported that anatase TiO<sub>2</sub> nanorods (TiO<sub>2</sub>-RMA) with the {010} facet modified with rutile TiO<sub>2</sub> NPs were synthesized *via* a one-step method.<sup>228</sup> The HRTEM image of TiO<sub>2</sub>-RMA is shown in Fig. 22a. The clear lattice fringes with a *d*-spacing of 0.32 nm and 0.35 nm for the rutile {110} and anatase {101} facets, respectively, were identified. Rutile-anatase heterojunction showed high CH<sub>4</sub> formation rate (2.36 μmol g<sup>-1</sup> h<sup>-1</sup>) under UV light irradiation, and the possible mechanism is shown in Fig. 22b. Due to the fact that the CB of anatase was more negative than that of rutile, e<sup>-</sup> could transfer from anatase to rutile. Therefore, e<sup>-</sup> and h<sup>+</sup> could be effectively separated. In a similar study, Gray *et al.* used the structure–function understanding to fabricate anatase–rutile TiO<sub>2</sub> phases for the photocatalytic reduction of CO<sub>2</sub>, and high selectivity toward CH<sub>4</sub> was obtained.<sup>229</sup> Gray *et al.* used the direct current (DC) magnetron sputtering to synthesize mixed-phase TiO<sub>2</sub> films (70% anatase and 30% rutile).<sup>230</sup> In another report, Zhao *et al.* synthesized Au–Pd bimetallic metal loaded onto 3DOM TiO<sub>2</sub> with the anatase–rutile phases.<sup>231</sup> The Au<sub>3</sub>Pd<sub>1</sub>/3DOM TiO<sub>2</sub> exhibited the maximum CH<sub>4</sub> formation rate (18.5 μmol g<sup>-1</sup> h<sup>-1</sup>) because the Au–Pd bimetallic metal had a relatively low Fermi level than that of TiO<sub>2</sub> and the separation of e<sup>-</sup>/h<sup>+</sup> pairs was accelerated.

The anatase, brookite, and rutile phases of TiO<sub>2</sub> can be accurately tuned by the solution's pH. For example, Ling *et al.* synthesized anatase–brookite mixed phases *via* a facile hydrothermal method.<sup>232</sup> The TEM image of anatase–brookite TiO<sub>2</sub> is

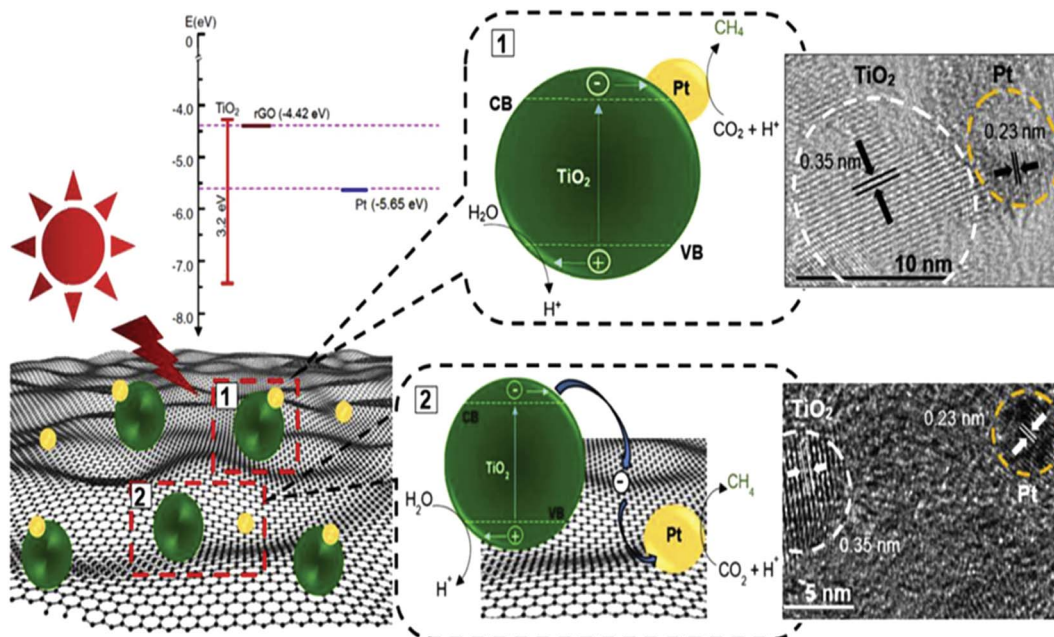


Fig. 21 Possible mechanism of Pt–GT composite. Reproduced from ref. 220 with permission from Elsevier.

shown in Fig. 23a. It was indicated that rod-like and spherical NPs were the typical structures of the brookite and anatase phases, respectively.<sup>233</sup> Anatase–brookite TiO<sub>2</sub> exhibited the highest CH<sub>3</sub>OH formation rate under both UV-vis and visible light irradiations. The charge transfer between anatase–brookite TiO<sub>2</sub> is shown in Fig. 23b. Here, e<sup>-</sup> could transfer from the CB of brookite to anatase to react with CO<sub>2</sub> and H<sup>+</sup> for the production of CH<sub>3</sub>OH, and h<sup>+</sup> could transfer from the VB of anatase to brookite to react with H<sub>2</sub>O to produce O<sub>2</sub> and H<sup>+</sup>. Li *et al.* also reported that the anatase–brookite mixed-phase TiO<sub>2</sub> showed the highest CO yield for the photocatalytic reduction of CO<sub>2</sub> when compared with single-phase anatase and brookite TiO<sub>2</sub> and P25.<sup>234</sup>

**4.4.2 Facet heterojunction.** Recent studies have confirmed that exposed facets play a critical role in improving the photocatalytic performance. On account of the first-principles calculations, the {101} facet of anatase TiO<sub>2</sub> is the primary CO<sub>2</sub> adsorption site for CO<sub>2</sub> reduction.<sup>235</sup> Moreover, the {101} facet can accelerate e<sup>-</sup> transfer from the surface of TiO<sub>2</sub> to CO<sub>2</sub>. Yu *et al.* first controlled the ratio of coexposed {001} and {101} facets of anatase TiO<sub>2</sub> for the photocatalytic reduction of CO<sub>2</sub> to CH<sub>4</sub>.<sup>235</sup> Fig. 24a shows the schematic involving the tuning of the {001} and {101} facets of anatase TiO<sub>2</sub> by adding HF during the preparation process. At the optimal facet ratio of {101} to {001} with 45 to 55 (adding 4.5 mL HF), TiO<sub>2</sub> exhibited the maximum CH<sub>4</sub> formation rate (1.35 μmol g<sup>-1</sup> h<sup>-1</sup>). The schematic of {001}

Table 6 Comparison of the typical phase or facet TiO<sub>2</sub> heterojunction systems

| Photocatalyst                      | Solvent                                             | Light source    | Time <sup>a</sup> | Formation rate <sup>b</sup>        | QE <sup>c</sup> (%)                    | Reference |
|------------------------------------|-----------------------------------------------------|-----------------|-------------------|------------------------------------|----------------------------------------|-----------|
| <b>Phase heterojunction</b>        |                                                     |                 |                   |                                    |                                        |           |
| TiO <sub>2</sub> –RMA              | H <sub>2</sub> O                                    | Hg lamp         | 8                 | CH <sub>4</sub> : 2.36             | —                                      | 228       |
| Mixed-phase TiO <sub>2</sub>       | Sodium bicarbonate + H <sub>2</sub> O + isopropanol | Hg lamp         | 3                 | CH <sub>4</sub> : 34.67            | —                                      | 229       |
| Mixed-phase TiO <sub>2</sub>       | H <sub>2</sub> O                                    | Hg lamp         | 4                 | CH <sub>4</sub> : 26.79            | —                                      | 225       |
| AuPd/3DOM–TiO <sub>2</sub>         | H <sub>2</sub> O                                    | Xe lamp         | 6                 | CH <sub>4</sub> : 18.5             | 0.41                                   | 231       |
| Anatase–brookite TiO <sub>2</sub>  | H <sub>2</sub> O                                    | Xe lamp         | 9                 | CH <sub>3</sub> OH: 0.478          | 0.0717                                 | 232       |
| Anatase–brookite TiO <sub>2</sub>  | H <sub>2</sub> O                                    | Solar simulator | 4                 | CO: 2.1                            | —                                      | 234       |
| <b>Facet heterojunction</b>        |                                                     |                 |                   |                                    |                                        |           |
| {101}/(001) TiO <sub>2</sub>       | H <sub>2</sub> O                                    | Xe lamp         | —                 | CH <sub>4</sub> : 1.35             | —                                      | 235       |
| N–TiO <sub>2</sub> –{101}/(001)    | H <sub>2</sub> O                                    | Xe lamp         | 2                 | CH <sub>3</sub> OH: 0.36           | —                                      | 236       |
| G/TiO <sub>2</sub> –001/101        | H <sub>2</sub> O                                    | Xe lamp         | 4                 | CO: 70.8<br>CH <sub>4</sub> : 27.4 | CO: 0.0557<br>CH <sub>4</sub> : 0.0864 | 237       |
| Pt/TiO <sub>2</sub> –{001/101}@rGO | H <sub>2</sub> O                                    | Xe lamp         | 8                 | CH <sub>4</sub> : 41.3             | 1.93                                   | 238       |
| TiO <sub>2</sub> –{110}/(111)      | 0.08 M NaHCO <sub>3</sub>                           | Xe lamp         | 5                 | CH <sub>3</sub> OH: 0.569          | 0.00569                                | 239       |

<sup>a</sup> Irradiation time, h. <sup>b</sup> Formation rate, μmol g<sup>-1</sup> h<sup>-1</sup>. <sup>c</sup> Quantum efficiency (%).

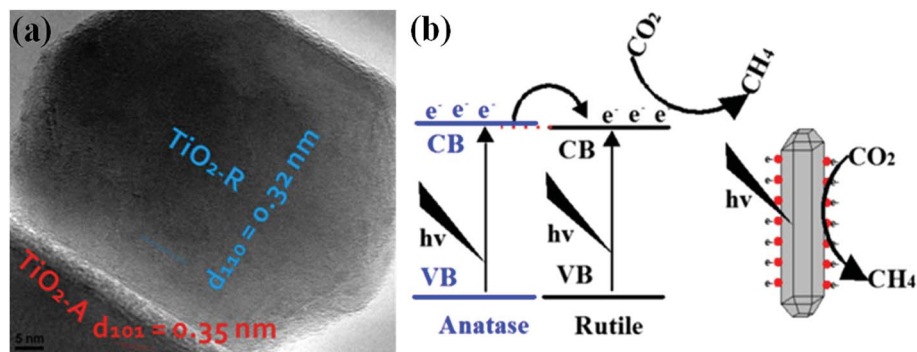


Fig. 22 (a) HRTEM image of TiO<sub>2</sub>-RMA; (b) possible mechanism of TiO<sub>2</sub>-RMA. Reproduced from ref. 228 with permission from Elsevier.

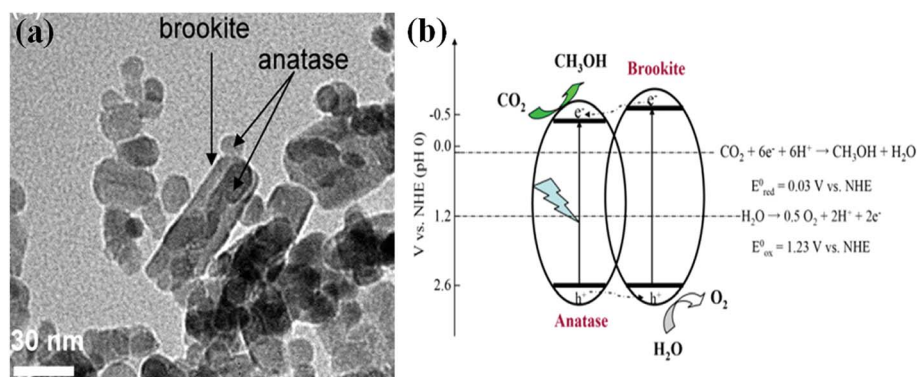


Fig. 23 (a) TEM image of anatase-brookite TiO<sub>2</sub>; (b) possible mechanism of anatase-brookite TiO<sub>2</sub>. Reproduced from ref. 232 with permission from Elsevier.

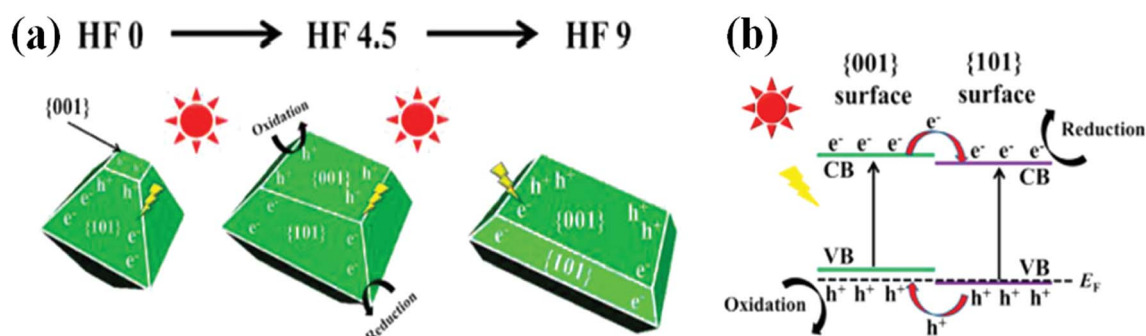


Fig. 24 (a) Schematic with tuning {001} and {101} facets of anatase TiO<sub>2</sub> by HF0, HF4.5, and HF9 samples; (b) charge transfer of {001} and {101} facets heterojunction of TiO<sub>2</sub>. Reproduced from ref. 235 with permission from the American Chemical Society.

and {101} facet heterojunction is shown in Fig. 24b. Under light irradiation, e<sup>-</sup> and h<sup>+</sup> could effectively transfer to the {101} and {001} facets, respectively. Therefore, {101} and {001} facets served as the reduction and oxidation activity sites, respectively. The {101} and {001} facets could form a facet heterojunction to enhance the photocatalytic performance. In another study, N-doped TiO<sub>2</sub> with 65% {001} and 35% {101} exposed facets obtained *via* a facile hydrothermal method was synthesized by Yu *et al.*<sup>236</sup> The improved CH<sub>3</sub>OH formation rate was ascribed to the synergistic effect between the facet heterojunction, N doping, and surface fluorination.

Due to the unique properties of GR, it is often used as a cocatalyst to enhance the photocatalytic activity. GR loaded on coexposed {001} and {101} facets of TiO<sub>2</sub> (GR/TiO<sub>2</sub>-001/101) was reported by Zhao *et al.*<sup>237</sup> The {001} and {101} facets could form the facet heterojunction, which, along with GR, could accelerate the separation of e<sup>-</sup> and h<sup>+</sup> to increase the CO yield. In a subsequent work, Zhao *et al.* synthesized Pt/TiO<sub>2</sub>@rGO core-shell-structured photocatalysts, where TiO<sub>2</sub> had 51% {001} and 49% {101} coexposed facets.<sup>238</sup> Pt NPs could gather and transfer e<sup>-</sup> from TiO<sub>2</sub> and then transfer to rGO, resulting in high charge separation efficiency. The addition of cocatalysts of Pt NPs and

rGO could lead to a high  $\text{CH}_4$  formation rate and quantum efficiency. Recently, Truong *et al.* found that the exposed high-index  $\{311\}$  facets of rutile  $\text{TiO}_2$  nanocrystals comprised two  $\{110\}$  terraces and one  $\{111\}$  step.<sup>239</sup> Each branched nanocrystal was bound with four facets of the high-index  $\{331\}$  facet. It was interesting to note that the flower-like particles had a higher  $\text{CH}_3\text{OH}$  yield ( $0.569 \mu\text{mol g}^{-1} \text{h}^{-1}$ ) than that of rod-like particles ( $0.388 \mu\text{mol g}^{-1} \text{h}^{-1}$ ) for the photocatalytic reduction of  $\text{CO}_2$ .

#### 4.5 Other $\text{TiO}_2$ -based heterojunctions

Dye molecules are usually chemisorbed on the surface of  $\text{TiO}_2$  for sensitization. Kang *et al.* reported the TDQD [(*E*)-3-(5-(4-(5-(4-(bis(4-(hexyloxy)phenyl)amino)phenyl)thiophen-2-yl)-6,7-diethyl-[1,2,5]thiadiazolo[3,4-*g*]quinoxalin-9-yl)thiophen-2-yl)-2-cyanoacrylic acid] dye loaded onto commercialized  $\text{TiO}_2$  (P25) to extend the visible light absorption region.<sup>240</sup> It exhibited a high  $\text{CH}_4$  yield of about  $1290 \mu\text{mol g}^{-1} \text{L}^{-1}$  after 8 h of light irradiation. Metal-organic frameworks (MOFs) are promising materials for  $\text{CO}_2$  reduction due to their high surface area to enhance  $\text{CO}_2$  adsorption. Petit *et al.* fabricated  $\text{NH}_2$ -UiO-66/ $\text{TiO}_2$  nanosheet heterojunctions *via* an *in situ* growth method for the first time.<sup>241</sup> The heterostructure could offer a high surface area and more active sites. Moreover, the large  $\text{CO}_2$  adsorption capacity could improve the  $\text{CO}_2$  level on the surface of  $\text{TiO}_2$  to react with the active sites to produce CO. Recently, hierarchical P25@microporous Co-Al-layered double hydroxides (Co-Al-LDHs) heterojunction for the photocatalytic reduction of  $\text{CO}_2$  with high activity and selectivity to produce CO was also described.<sup>242</sup> Here,  $\text{e}^-$  could transfer from the Co-Al-LDH to P25; in contrast,  $\text{h}^+$  could transfer from P25 to Co-Al-LDH. Therefore,  $\text{e}^-$  and  $\text{h}^+$  were effectively separated. This may offer a facile method to design and fabricate LDH-based materials for energy conversion and pollutant degradation.

## 5. Conclusions and perspectives

In summary, this review discusses the design and fabrication of  $\text{TiO}_2$ -based heterojunction photocatalysts for the photocatalytic reduction of  $\text{CO}_2$  into solar fuels. The photocatalytic reduction of  $\text{CO}_2$  into solar fuels is a promising method to overcome energy crises and environmental problems. However, the photocatalytic reduction of  $\text{CO}_2$  is far away from industrial applications.<sup>102,243–245</sup> On one hand, there are some common bottlenecks in this process. The energy efficiency that evaluates the transformation of photon energy into chemical energy is very low. For example, Chen *et al.* reported the energy efficiency was 1.158% by  $\text{NiO}/\text{InTaO}_4$  under visible light illumination.<sup>246</sup> Furthermore, the instantaneous oxidation of the formed hydrocarbons may occur during  $\text{CO}_2$  reduction; therefore, we should minimize hydrocarbon conversion under process conditions needed for  $\text{CO}_2$  reduction.<sup>247</sup> In addition, hydrocarbons might be retained on the surface of the catalysts during the synthesis processes; therefore, we should eliminate the impact as much as possible. On the other hand, the selectivity of the desired products is low because the HER reaction is accompanied by  $\text{CO}_2$  reduction, and it is a rather complex

process.<sup>248</sup>  $\text{CO}$ ,  $\text{CH}_4$ ,  $\text{HCOOH}$ ,  $\text{CH}_3\text{OH}$ ,  $\text{HCHO}$ ,  $\text{C}_2\text{H}_4$ , and  $\text{CH}_3\text{-CH}_2\text{OH}$  can be observed in some catalytic systems. The adsorption and activation of  $\text{CO}_2$  onto the surface of catalysts can play some important roles in improving the selectivity toward the desired products. The increase in the catalyst surface area, introduction of surface defects (*e.g.*, oxygen and sulfur vacancies), and noble-metal cocatalysts have been widely selected strategies to enhance the adsorption and activation of  $\text{CO}_2$ .<sup>249</sup> The mechanism of  $\text{CO}_2$  reduction should be comprehensively investigated in future studies. At the molecular level through *in situ* characterization techniques, we can better understand the reaction pathways of  $\text{CO}_2$  reduction. The reaction barriers and rate-determining steps of  $\text{CO}_2$  reduction have an important effect to enhance the conversion efficiency.

Developing efficient photocatalysts is the key for improving the photocatalytic reduction of  $\text{CO}_2$ . The attractive properties of heterojunction photocatalysts can be used to increase the charge separation and suppress the recombination of  $\text{e}^-/\text{h}^+$  pairs. Therefore, designing a heterojunction is an effective strategy for high-activity photocatalysts in the photocatalytic reduction of  $\text{CO}_2$ . Currently,  $\text{TiO}_2$ -based heterojunction photocatalysts for the photocatalytic reduction of  $\text{CO}_2$  are still in their early stages of development, and we believe that there is room for developing novel, powerful photocatalysts for  $\text{CO}_2$  reduction. Interfacial or heterojunction quantitative control, facet control, surface decorations, band structure alignment, and interfacial *in situ* characterization on the heterojunction photocatalysts can also be critically important toward efficient photocatalysis, which should be further investigated.

At the same time, other strategies for the photocatalytic reduction of  $\text{CO}_2$  are of equal importance. First, the adsorption of  $\text{CO}_2$  over the surface of the photocatalyst is a very important step for the photocatalytic reduction of  $\text{CO}_2$ . The basic metal oxide can increase the adsorption of  $\text{CO}_2$  and it can serve as a cocatalyst for contributing toward the photocatalytic reduction of  $\text{CO}_2$ . Second, the morphology has an important effect on the photocatalytic performance. It influences the surface structure, which can control the energy band position and surface reactivity, and affects the transfer of photogenerated carriers. Third, the size of the photocatalyst particles also influences the photocatalytic activity. The optimum particle size of the photocatalyst affects the charge-carrier dynamics and light-harvesting efficiency. A decrease in the particle size may increase the bandgap energy and decrease the light-harvesting efficiency. Fourth, the reductive half-reaction of  $\text{CO}_2$  is coupled with the oxidative half-reaction of water oxidation to  $\text{O}_2$ . Further, more attention should be paid to the other half-reaction of water oxidation in future studies. In order to enhance the activity of  $\text{CO}_2$  reduction, it can accelerate the oxidation reaction by adopting certain typical strategies for enhanced  $\text{O}_2$  evolution in solar water splitting, such as adding some oxidizing auxiliaries.

Moreover, we also can develop other new types of photo-reactors for the reduction of  $\text{CO}_2$  to enhance the photocatalytic activity. Therefore, all the factors should be taken into account in the process of design and fabrication of the photocatalytic reduction of  $\text{CO}_2$  into solar fuels. We hope that this review can

stimulate new insights and provide guidelines for future research at the intersection of the photocatalytic reduction of CO<sub>2</sub> into solar fuels, which can benefit humankind.

## Conflicts of interest

There are no conflicts to declare.

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## References

- J. Peng, L. Lu and H. Yang, *Renewable Sustainable Energy Rev.*, 2013, **19**, 255–274.
- B. Khezri, A. C. Fisher and M. Pumera, *J. Mater. Chem. A*, 2017, **5**, 8230–8246.
- J. Zhao, X. Wang, Z. Xu and J. S. C. Loo, *J. Mater. Chem. A*, 2014, **2**, 15228–15233.
- A. Álvarez, A. Bansode, A. Urakawa, A. V. Bavykina, T. A. Wezendonk, M. Makkee, J. Gascon and F. Kapteijn, *Chem. Rev.*, 2017, **117**, 9804–9838.
- O. Ola and M. M. Maroto-Valer, *J. Photochem. Photobiol., C*, 2015, **24**, 16–42.
- R. Shi, G. I. N. Waterhouse and T. Zhang, *Sol. RRL*, 2017, **1**, 1700126.
- N. Pegiosa, V. Bliznukb, S. Prüntec, J. M. Schneiderc, R. Palkovits and K. Simeonov, *RSC Adv.*, 2018, **8**, 606–618.
- G. Gao, Y. Jiao, E. R. Waclawik and A. Du, *J. Am. Chem. Soc.*, 2016, **138**, 6292–6297.
- G. Yin, M. Nishikawa, Y. Nosaka, N. Srinivasan, D. Atarashi, E. Sakai and M. Miyauchi, *ACS Nano*, 2015, **9**, 2111–2119.
- W. Wang, W. An, B. Ramalingam, S. Mukherjee, D. M. Niedzwiedzki, S. Gangopadhyay and P. Biswas, *J. Am. Chem. Soc.*, 2012, **134**, 11276–11281.
- Y. Zhu, Z. Xu, W. Jiang, S. Zhong, L. Zhao and S. Bai, *J. Mater. Chem. A*, 2017, **5**, 2619–2628.
- K. H. Kim, S. Kim, B. C. Moon, J. W. Choi, H. M. Jeong, Y. Kwon, S. Kwon, H. S. Choib and J. K. Kang, *J. Mater. Chem. A*, 2017, **5**, 8274–8279.
- G. R. Dey, A. D. Belapurkar and K. Kishore, *J. Photochem. Photobiol., A*, 2004, **163**, 503–508.
- R. Kuriki, H. Matsunaga, T. Nakashima, K. Wada, A. Yamakata, O. Ishitani and K. Maeda, *J. Am. Chem. Soc.*, 2016, **138**, 5159–5170.
- S. Kaneco, Y. Ueno, H. Katsumata, T. Suzuki and K. Ohta, *Chem. Eng. J.*, 2006, **119**, 107–112.
- L. Roy, P. M. Zimmerman and A. Paul, *Chem.–Eur. J.*, 2011, **17**, 435–439.
- Q. Shen, X. Huang, J. Liu, C. Guo and G. Zhao, *Appl. Catal., B*, 2017, **201**, 70–76.
- H. Zhou, T. Fan and D. Zhang, *ChemCatChem*, 2011, **3**, 513–528.
- X. Huang, Q. Shen, J. Liu, N. Yang and G. Zhao, *Energy Environ. Sci.*, 2016, **9**, 3161–3171.
- S. Wang, B. Guan and X. W. D. Lou, *Energy Environ. Sci.*, 2018, **11**, 306–310.
- T. Inoue, A. Fujishima, S. Konishi and K. Honda, *Nature*, 1979, **277**, 637–638.
- I. Hashemizadeh, V. B. Golovko, J. Choi, D. C. W. Tsang and A. C. K. Yip, *Chem. Eng. J.*, 2018, **347**, 64–73.
- C. Dong, C. Lian, S. Hu, Z. Deng, J. Gong, M. Li, H. Liu, M. Xing and J. Zhang, *Nat. Commun.*, 2018, **9**, 1252.
- B. Yu, Y. Zhou, P. Li, W. Tu, P. Li, L. Tang, J. Ye and Z. Zou, *Nanoscale*, 2016, **8**, 11870–11874.
- Y. Ma, X. Wang, Y. Jia, X. Chen, H. Han and C. Li, *Chem. Rev.*, 2014, **114**, 9987–10043.
- A. Nikokavoura and C. Trapalis, *Appl. Surf. Sci.*, 2017, **391**, 149–174.
- H. Li, Y. Gao, Z. Xiong, C. Liao and K. Shih, *Appl. Surf. Sci.*, 2018, **439**, 552–559.
- D. Xu, B. Cheng, W. Wang, C. Jiang and J. Yu, *Appl. Catal., B*, 2018, **231**, 368–380.
- L. Lin, C. Hou, X. Zhang, Y. Wang, Y. Chen and T. He, *Appl. Catal., B*, 2018, **221**, 312–319.
- Z. Jiang, W. Wan, H. Li, S. Yuan, H. Zhao and P. Wong, *Adv. Mater.*, 2018, **30**, 1706108.
- A. Kumar, A. Kumar, G. Sharma, A. H. Al-Muhtaseb, M. Naushad, A. A. Ghfar, C. S. Guo and F. J. Stadler, *Chem. Eng. J.*, 2018, **339**, 393–410.
- X. Jiao, Z. Chen, X. Li, Y. Sun, S. Gao, W. Yan, C. Wang, Q. Zhang, Y. Lin, Y. Luo and Y. Xie, *J. Am. Chem. Soc.*, 2017, **139**, 7586–7594.
- S. Wang, B. Guan and X. W. D. Lou, *J. Am. Chem. Soc.*, 2018, **140**, 5037–5040.
- J. Chen, F. Xin, X. Yin, T. Xiang and Y. Wang, *RSC Adv.*, 2015, **5**, 3833–3839.
- G. Yang, D. Chen, H. Ding, J. Feng, J. Zhang, Y. Zhu, S. Hamid and D. W. Bahnemann, *Appl. Catal., B*, 2017, **219**, 611–618.
- L. Xiao, R. Lin, J. Wang, C. Cui, J. Wang and Z. Li, *J. Colloid Interface Sci.*, 2018, **523**, 151–158.
- S. Cao, B. Shen, T. Tong, J. Fu and J. Yu, *Adv. Funct. Mater.*, 2018, 1800136.
- Z. Jiang, X. Liang, H. Zheng, Y. Liu, Z. Wang, P. Wang, X. Zhang, X. Qin, Y. Dai, M. H. Whangbo and B. Huang, *Appl. Catal., B*, 2017, **219**, 209–215.
- M. Wang, Q. Han, L. Li, L. Tang, H. Li, Y. Zhou and Z. Zou, *Nanotechnology*, 2017, **28**, 1–8.
- H. Cheng, B. Huang, Y. Liu, Z. Wang, X. Qin, X. Zhang and Y. Dai, *Chem. Commun.*, 2012, **48**, 9729–9731.
- X. Kong, Y. Y. Choo, S. Chai, A. K. Soh and A. R. Mohamed, *Chem. Commun.*, 2016, **52**, 14242–14245.

- 42 P. Wang, Y. Bai, P. Luo and J. Liu, *Catal. Commun.*, 2013, **38**, 82–85.
- 43 Q. Li, B. Guo, J. Yu, J. Ran, B. Zhang, H. Yan and J. Gong, *J. Am. Chem. Soc.*, 2011, **133**, 10878–10884.
- 44 X. Pan and Y. Xu, *J. Phys. Chem. C*, 2015, **119**, 7184–7194.
- 45 W. Tu, Y. Zhou, Q. Liu, S. Yan, S. Bao, X. Wang, M. Xiao and Z. Zou, *Adv. Funct. Mater.*, 2013, **23**, 1743–1749.
- 46 Y. Liang, B. K. Vijayan, K. A. Gray and M. C. Hersam, *Nano Lett.*, 2011, **11**, 2865–2870.
- 47 S. Ijaz, M. F. Ehsan, M. N. Ashiq, N. Karamat and T. He, *Appl. Surf. Sci.*, 2016, **390**, 550–559.
- 48 Y. Wei, J. Jiao, Z. Zhen, J. Liu, J. Li, G. Jiang, Y. Wang and A. Duan, *Appl. Catal., B*, 2015, **179**, 422–432.
- 49 S. Wang and X. Wang, *Appl. Catal., B*, 2015, **162**, 494–500.
- 50 P. Praus, O. Kozák, K. Kočí, A. Panáček and R. Dvorský, *J. Colloid Interface Sci.*, 2011, **360**, 574–579.
- 51 S. Xie, Y. Wang, Q. Zhang, W. Deng and Y. Wang, *Chem. Commun.*, 2015, **51**, 3430–3433.
- 52 Y. He, Y. Wang, L. Zhang, B. Teng and M. Fan, *Appl. Catal., B*, 2015, **168–169**, 1–8.
- 53 Q. Guo, Q. Zhang, H. Wang and Z. Zhao, *Catal. Commun.*, 2018, **103**, 24–28.
- 54 S. R. Lingampalli, M. M. Ayyub, G. Magesh and C. N. R. Rao, *Chem. Phys. Lett.*, 2018, **691**, 28–32.
- 55 K. Li, B. Peng and T. Peng, *ACS Catal.*, 2016, **6**, 7485–7527.
- 56 W. Yuan, L. Cheng, Y. An, S. Lv, H. Wu, X. Fan, Y. Zhang, X. Guo and J. Tang, *Adv. Sci.*, 2018, 1700870.
- 57 M. Ge, C. Cao, J. Huang, S. Li, Z. Chen, K. Zhang, S. S. Al-Deyabd and Y. Lai, *J. Mater. Chem. A*, 2016, **4**, 6772–6801.
- 58 J. Pan, X. Wu, L. Wang, G. Liu, G. Q. M. Lu and H. Cheng, *Chem. Commun.*, 2011, **47**, 8361–8363.
- 59 J. F. d. Brito and M. V. B. Zaroni, *Chem. Eng. J.*, 2017, **318**, 264–271.
- 60 J. Low, B. Cheng and J. Yu, *Appl. Surf. Sci.*, 2017, **392**, 658–686.
- 61 S. Xie, Y. Wang, Q. Zhang, W. Deng and Y. Wang, *ACS Catal.*, 2014, **4**, 3644–3653.
- 62 S. Xie, Y. Wang, Q. Zhang, W. Fan, W. Deng and Y. Wang, *Chem. Commun.*, 2013, **49**, 2451–2453.
- 63 G. Liu, S. Xie, Q. Zhang, Z. Tian and Y. Wang, *Chem. Commun.*, 2015, **51**, 13654–13657.
- 64 Q. Zhai, S. Xie, W. Fan, Q. Zhang, Y. Wang, W. Deng and Y. Wang, *Angew. Chem., Int. Ed.*, 2013, **52**, 5776–5779.
- 65 C. Yu, L. Wei, J. Chen, Y. Xie, W. Zhou and Q. Fan, *Ind. Eng. Chem. Res.*, 2014, **53**, 5759–5766.
- 66 C. Yu, L. Wei, W. Zhou, J. Chen, Q. Fan and H. Liu, *Appl. Surf. Sci.*, 2014, **319**, 312–318.
- 67 C. Yu, L. Wei, J. Chen, W. Zhou, Q. Fan and J. Yu, *Rare Met.*, 2016, **35**, 475–480.
- 68 C. Yu, J. Chen, W. Zhou, L. Wei and Q. Fan, *Mater. Res. Innovations*, 2015, **19**, 54–58.
- 69 J. Tian, Z. Zhao, A. Kumar, R. I. Boughton and H. Liu, *Chem. Soc. Rev.*, 2014, **43**, 6920–6937.
- 70 J. Tian, Z. Wu, Z. Liu, C. Yu, K. Yang, L. Zhu, W. Huang and Y. Zhou, *Chin. J. Catal.*, 2017, **38**, 1899–1908.
- 71 C. Yu, G. Li, S. Kumar, K. Yang and R. Jin, *Adv. Mater.*, 2014, **26**, 892–898.
- 72 J. Jin, J. Yu, D. Guo, C. Cui and W. Ho, *Small*, 2015, **11**, 5262–5271.
- 73 H. Li, X. Zhang and D. R. MacFarlane, *Adv. Energy Mater.*, 2015, **5**, 1401077.
- 74 X. Jiao, X. Li, X. Jin, Y. Sun, J. Xu, L. Liang, H. Ju, J. Zhu, Y. Pan, W. Yan, Y. Lin and Y. Xie, *J. Am. Chem. Soc.*, 2017, **49**, 18044–18051.
- 75 Z. Wang, K. Teramura, Z. Huang, S. Hosokawa, Y. Sakata and T. Tanaka, *Catal. Sci. Technol.*, 2016, **6**, 1025–1032.
- 76 H. Shi, G. Chen, C. Zhang and Z. Zou, *ACS Catal.*, 2014, **4**, 3637–3643.
- 77 Y. He, L. Zhang, M. Fan, X. Wang, M. L. Walbridge, Q. Nong, Y. Wu and L. Zhao, *Sol. Energy Mater. Sol. Cells*, 2015, **137**, 175–184.
- 78 T. W. Kim, I. Y. Kim, T. S. Jung, C. H. Ko and S. J. Hwang, *Adv. Funct. Mater.*, 2013, **23**, 4377–4385.
- 79 G. Bucher and W. Sander, *Science*, 2014, **346**, 544–545.
- 80 S. Nasri, Y. Ajili, N. E. Jaidane, Y. N. Kalugina, P. Halvick, T. Stoecklin and M. Hochlaf, *J. Chem. Phys.*, 2015, **142**, 174301–174308.
- 81 J. R. Bolton, *Science*, 1978, **202**, 705–711.
- 82 J. M. Lehn and R. Ziessel, *Proc. Natl. Acad. Sci. U. S. A.*, 1982, **79**, 701–704.
- 83 I. Willner, R. Maidan, D. Mandler, H. Dum, G. Dorr and K. Zengerlet, *J. Am. Chem. Soc.*, 1987, **109**, 6080–6086.
- 84 L. Guo, Y. Wang and T. He, *Chem. Rec.*, 2016, **16**, 1918–1933.
- 85 V. P. Indrakanti, J. D. Kubickib and H. H. Schobert, *Energy Environ. Sci.*, 2009, **2**, 745–758.
- 86 B. Kumar, M. Llorente, J. Froehlich, T. Dang, A. Sathrum and C. P. Kubiak, *Annu. Rev. Phys. Chem.*, 2012, **63**, 541–569.
- 87 A. J. Morris, G. J. Meyer and E. Fujita, *Acc. Chem. Res.*, 2009, **42**, 1983–1994.
- 88 E. E. Benson, C. P. Kubiak, A. J. Sathrum and J. M. Smieja, *Chem. Soc. Rev.*, 2009, **38**, 89–99.
- 89 A. H. Yahaya, M. A. Gondal and A. Hameed, *Chem. Phys. Lett.*, 2004, **400**, 206–212.
- 90 A. D. Handoko, K. Li and J. W. Tang, *Curr. Opin. Chem. Eng.*, 2012, **2**, 200–206.
- 91 K. Maeda and K. Domen, *J. Phys. Chem. C*, 2007, **111**, 7851–7861.
- 92 O. Carp, C. L. Huisman and A. Reller, *Prog. Solid State Chem.*, 2004, **32**, 33–177.
- 93 U. G. Akpan and B. H. Hameed, *J. Hazard. Mater.*, 2009, **170**, 520–529.
- 94 K. Koci, L. Obalova and Z. Lacny, *Chem. Pap.*, 2008, **62**, 1–9.
- 95 A. Dhakshinamoorthy, S. Navalon, A. Corma and H. Garcia, *Energy Environ. Sci.*, 2012, **5**, 9217–9233.
- 96 J. Chen, F. Xin, S. Qin and X. Yin, *Chem. Eng. J.*, 2013, **230**, 506–512.
- 97 G. Zeng, J. Qiu, Z. Li, P. Pavaskar and S. B. Cronin, *ACS Catal.*, 2014, **4**, 3512–3516.
- 98 M. Gui, S. Chai and A. R. Mohamed, *Appl. Surf. Sci.*, 2014, **319**, 37–43.
- 99 L. Yuan and Y. Xu, *Appl. Surf. Sci.*, 2015, **342**, 154–167.
- 100 S. Rani, N. Bao and S. C. Roy, *Appl. Surf. Sci.*, 2014, **289**, 203–208.

- 101 S. Xie, Q. Zhang, G. Liu and Y. Wang, *Chem. Commun.*, 2016, **52**, 35–59.
- 102 S. N. Habisreutinger, L. Schmidt-Mende and J. K. Stolarczyk, *Angew. Chem., Int. Ed.*, 2013, **52**, 7372–7408.
- 103 X. Li, J. Wen, J. Low, Y. Fang and J. Yu, *Sci. China Mater.*, 2014, **57**, 70–100.
- 104 J. L. White, M. F. Baruch, J. E. Pander, Y. Hu, I. C. Fortmeyer, J. E. Park, T. Zhang, K. Liao, J. Gu, Y. Yan, T. W. Shaw, E. Abelev and A. B. Bocarsly, *Chem. Rev.*, 2015, **115**, 12888–12935.
- 105 A. L. Linsebigler, G. Lu and J. T. Yates, *Chem. Rev.*, 1995, **95**, 735–758.
- 106 H. Jiao, X. Yu, Z. Liu, P. Kuang and Y. Zhang, *RSC Adv.*, 2015, **5**, 16239–16249.
- 107 R. Chen, S. Pang, H. An, J. Zhu, S. Ye, Y. Gao, F. Fan and C. Li, *Nat. Energy*, 2018, **3**, 655–663.
- 108 Z. T. Hu, J. Liu, X. Yan, W. D. Oh and T. T. Lim, *Chem. Eng. J.*, 2015, **262**, 1022–1032.
- 109 Z. Y. Asu, M. Yoshiteru, M. Yasuaki and H. I. Nilsun, *Appl. Catal., B*, 2015, **172–173**, 7–17.
- 110 G. Liu, L. Wang, H. Yang, H. Cheng and G. Lu, *J. Mater. Chem.*, 2010, **20**, 831–843.
- 111 L. Zhang, Y. Li, Q. Zhang and H. Wang, *CrystEngComm*, 2013, **15**, 5986–5993.
- 112 L. Wang, J. Ge, A. Wang, M. Deng, X. Wang, S. Bai, R. Li, J. Jiang, Q. Zhang, Y. Luo and Y. Xiong, *Angew. Chem.*, 2014, **53**, 5107–5111.
- 113 Y. Jia, S. Shen, D. Wang, X. Wang, J. Shi, F. Zhang, H. Han and C. Li, *J. Mater. Chem. A*, 2013, **1**, 7905–7912.
- 114 Z. Gan, X. Wu, M. Meng, X. Zhu, L. Yang and P. Chu, *ACS Nano*, 2014, **8**, 9304–9310.
- 115 Y. Gu, M. Xing and J. Zhang, *Appl. Surf. Sci.*, 2014, **319**, 8–15.
- 116 M. Bideau, B. Claudel, C. Dubien, L. Faure and H. Kazouan, *J. Photochem. Photobiol., A*, 1995, **91**, 137–144.
- 117 S. N. Habisreutinger, L. Schmidt-Mende and J. K. Stolarczyk, *Angew. Chem., Int. Ed.*, 2013, **52**, 7372–7408.
- 118 A. Corma and H. Garcia, *J. Catal.*, 2013, **308**, 168–175.
- 119 Z. Xiong, Z. Lei, B. Gong, X. Chen, Y. Zhao, J. Zhang, C. Zheng and J. C. S. Wu, *Catal. Commun.*, 2017, **89**, 4–8.
- 120 V. H. Nguyen and J. C. S. Wu, *Appl. Catal., A*, 2018, **550**, 122–141.
- 121 H. Wu, N. H. Nguyen, H. Bai, S. Chang and J. C. S. Wu, *RSC Adv.*, 2015, **5**, 63142–63151.
- 122 O. Ola, M. Maroto-Valer, D. Liu, S. Mackintosh, C. W. Lee and J. C. S. Wu, *Appl. Catal., B*, 2012, **126**, 172–179.
- 123 W. H. Lee, C. H. Liao, M. F. Tsai, C. W. Huang and J. C. S. Wu, *Appl. Catal., B*, 2013, **132–133**, 445–451.
- 124 Z. Y. Wang, H. C. Chou, J. C. S. Wu, D. P. Tsai and G. Mul, *Appl. Catal., A*, 2010, **380**, 172–177.
- 125 J. C. S. Wu, H. M. Lin and C. L. Lai, *Appl. Catal., A*, 2005, **296**, 194–200.
- 126 J. C. S. Wu, T. H. Wu, T. Chu, H. Huang and D. Tsai, *Top. Catal.*, 2008, **47**, 131–136.
- 127 T. V. Nguyen and J. C. S. Wu, *Appl. Catal., A*, 2008, **335**, 112–120.
- 128 K. Yuan, L. Yang, X. Du and Y. Yang, *Energy Convers. Manage.*, 2014, **81**, 98–105.
- 129 P. Usubharatana, D. McMartin, A. Veawab and P. Tontiwachwuthikul, *Ind. Eng. Chem. Res.*, 2006, **45**, 2558–2568.
- 130 P. Y. Liou, S. C. Chen, J. C. S. Wu, D. Liu, S. Mackintosh, M. Maroto-Valer and R. Linforth, *Energy Environ. Sci.*, 2011, **4**, 1487–1494.
- 131 M. Tahir and N. S. Amin, *Chem. Eng. J.*, 2013, **230**, 314–327.
- 132 M. Tahir and B. Tahir, *Appl. Surf. Sci.*, 2016, **377**, 244–252.
- 133 X. Cheng, R. Chen, X. Zhu, Q. Liao, L. An, D. Ye, X. He, S. Li and L. Li, *Energy*, 2017, **120**, 276–282.
- 134 M. Cheng, S. Yang, R. Chen, X. Zhu, Q. Liao and Y. Huang, *Int. J. Hydrogen Energy*, 2017, **42**, 9722–9732.
- 135 X. Cheng, R. Chen, X. Zhu, Q. Liao, X. He, S. Li and L. Li, *Int. J. Hydrogen Energy*, 2016, **41**, 2457–2465.
- 136 H. Wang, L. Zhang, Z. Chen, J. Hu, S. Li, Z. Wang, J. Liu and X. Wang, *Chem. Soc. Rev.*, 2014, **43**, 5234–5244.
- 137 P. Zhou, J. Yu and M. Jaroniec, *Adv. Mater.*, 2014, **26**, 4920–4935.
- 138 S. Bai, J. Jiang, Q. Zhang and Y. Xiong, *Chem. Soc. Rev.*, 2015, **44**, 2893–2939.
- 139 K. Maeda, *ACS Catal.*, 2013, **3**, 1486–1503.
- 140 D. Zeng, K. Yang, C. Yu, F. Chen, X. Li, Z. Wu and H. Liu, *Appl. Catal., B*, 2018, **237**, 449–463.
- 141 C. C. Yang, Y. H. Yu, B. van der Linden, J. C. S. Wu and G. Mul, *J. Am. Chem. Soc.*, 2010, **132**, 8398–8406.
- 142 L. Yu, G. Li, X. Zhang, X. Ba, G. Shi, Y. Li, P. K. Wong, J. Yu and Y. Yu, *ACS Catal.*, 2016, **6**, 6444–6454.
- 143 X. An, K. Li and J. Tang, *ChemSusChem*, 2014, **7**, 1086–1093.
- 144 M. L. Ovcharov, A. M. Mishura, N. D. Shcherban, S. M. Filonenko and V. M. Granchak, *Sol. Energy*, 2016, **139**, 452–457.
- 145 H. Xu, S. Ouyang, L. Liu, D. Wang, T. Kako and J. Ye, *Nanotechnology*, 2014, **25**, 165402.
- 146 F. Bi, M. F. Ehsan, W. Liu and T. He, *Chin. J. Chem.*, 2015, **33**, 112–118.
- 147 D. Liu, Y. Fernández, O. Ola, S. Mackintosh, M. Maroto-Valer, C. M. A. Parlett, A. F. Lee and J. C. S. Wu, *Catal. Commun.*, 2012, **25**, 78–82.
- 148 M. A. M. Júnior, A. Morais and A. F. Nogueira, *Microporous Mesoporous Mater.*, 2016, **234**, 1–11.
- 149 J. Wang, G. Ji, Y. Liu, M. A. Gondal and X. Chang, *Catal. Commun.*, 2014, **46**, 17–21.
- 150 Y. Li, W. Zhang, X. Shen, P. Peng, L. Xiong and Y. Yu, *Chin. J. Catal.*, 2015, **36**, 2229–2236.
- 151 H. R. Kim, A. Razzaq, C. A. Grimes and S. I. In, *J. CO<sub>2</sub> Util.*, 2017, **20**, 91–96.
- 152 Q. Zhang, L. Huang, S. Kang, C. Yin, Z. Ma, L. Cui and Y. Wang, *RSC Adv.*, 2017, **7**, 43642–43647.
- 153 R. Gusain, P. Kumar, O. P. Sharma, S. L. Jain and O. P. Khatri, *Appl. Catal., B*, 2016, **181**, 352–362.
- 154 M. H. Razali and M. Yusoff, *Mater. Lett.*, 2018, **221**, 168–171.
- 155 M. Akbari and S. Sharifnia, *Mater. Lett.*, 2017, **194**, 110–113.
- 156 A. Yarahmadi and S. Sharifnia, *Dyes Pigm.*, 2014, **107**, 140–145.
- 157 S. Qin, F. Xin, Y. Liu, X. Yin and W. Ma, *J. Colloid Interface Sci.*, 2011, **356**, 257–261.



- 158 S. Slamet, H. W. Nasution, E. Purnama, S. Kosela and J. Gunlazuardi, *Catal. Commun.*, 2005, **6**, 313–319.
- 159 B. Fang, Y. Xing, A. Bonakdarpour, S. Zhang and D. P. Wilkinson, *ACS Sustainable Chem. Eng.*, 2015, **3**, 2381–2388.
- 160 S. M. Park, A. Razzaq, Y. H. Park, S. Sorcar, Y. Park, C. A. Grimes and S. I. In, *ACS Omega*, 2016, **1**, 868–875.
- 161 S. I. In, D. D. Vaughn II and R. E. Schaak, *Angew. Chem.*, 2012, **124**, 3981–3984.
- 162 Z. Zhang and J. Li, *J. Mater. Sci.*, 2011, **46**, 3590–3596.
- 163 G. Marci, E. I. García-López and L. Palmisano, *Catal. Commun.*, 2014, **53**, 38–41.
- 164 Y. Zou, X. Wang, Y. Ai, Y. Liu, Y. Ji, H. Wang, T. Hayat, A. Alsaedi, W. Hu and X. Wang, *J. Mater. Chem. A*, 2016, **4**, 14170–14179.
- 165 Y. Pan, Z. Liu, W. Wang, C. Peng, K. Shi and X. Ji, *J. Mater. Chem. A*, 2016, **4**, 2537–2549.
- 166 N. R. Singha, M. Karmakar, M. Mahapatra, H. Mondal, A. Dutta, M. Deb, M. Mitra, C. Roy and P. K. Chattopadhyay, *J. Mater. Chem. A*, 2018, **6**, 8078–8100.
- 167 L. Cui, J. Wu and H. Ju, *ACS Appl. Mater. Interfaces*, 2014, **6**, 16210–16216.
- 168 C. Cao, C. Hua, W. Shen, S. Wang, Y. Tian and X. Wang, *J. Alloys Compd.*, 2012, **523**, 139–145.
- 169 G. Song, F. Xin, J. Chen and X. Yin, *Appl. Catal., A*, 2014, **473**, 90–95.
- 170 S. Ding, X. Yin, X. Lü, Y. Wang, F. Huang and D. Wan, *ACS Appl. Mater. Interfaces*, 2012, **4**, 306–311.
- 171 A. A. Beigi, S. Fatemi and Z. Salehi, *J. CO<sub>2</sub> Util.*, 2014, **7**, 23–29.
- 172 X. Li, H. Liu, D. Luo, J. Li, Y. Huang, H. Li, Y. Fang, Y. Xu and L. Zhu, *Chem. Eng. J.*, 2012, **180**, 151–158.
- 173 C. Wang, R. L. Thompson, P. Ohodnicki, J. Baltrus and C. Matranga, *J. Mater. Chem.*, 2011, **21**, 13452–13457.
- 174 J. Jiao, Y. Wei, Z. Zhao, J. Liu, J. Li, A. Duan and G. Jiang, *Ind. Eng. Chem. Res.*, 2014, **53**, 17345–17354.
- 175 Y. Wang, B. Li, C. Zhang, L. Cui, S. Kang, X. Li and L. Zhou, *Appl. Catal., B*, 2013, **130–131**, 277–284.
- 176 W. Yu, D. Xu and T. Peng, *J. Mater. Chem. A*, 2015, **3**, 19936–19947.
- 177 G. Xi, S. Ouyang and J. Ye, *Chem.–Eur. J.*, 2011, **17**, 9057–9061.
- 178 Z. Xiong, Z. Lei, S. Ma, X. Chen, B. Gong, Y. Zhao, J. Zhang, C. Zheng and J. C. S. Wu, *Appl. Catal., B*, 2017, **219**, 412–424.
- 179 Q. D. Truong, J. Liu, C. Chung and Y. Ling, *Catal. Commun.*, 2012, **19**, 85–89.
- 180 Q. Kang, T. Wang, P. Li, L. Liu, K. Chang, M. Li and J. Ye, *Angew. Chem.*, 2015, **127**, 855–859.
- 181 M. Reli, P. W. Huo, M. Šihor, N. Ambrožová, I. Troppová, L. Matějová, J. Lang, L. Svoboda, P. Kuśtrowski, M. Ritz, P. Praus and K. Kočí, *J. Phys. Chem. A*, 2016, **120**, 8564–8573.
- 182 S. Zhou, Y. Liu, J. Li, Y. Wang, G. Jiang, Z. Zhao, D. Wang, A. Duan, J. Liu and Y. Wei, *Appl. Catal., B*, 2014, **158–159**, 20–29.
- 183 M. A. Asi, C. He, M. H. Su, D. Xia, L. Lin, H. Deng, Y. Xiong, R. Qiu and X. Li, *Catal. Today*, 2011, **175**, 256–263.
- 184 H. Li, S. Gan, H. Wang, D. Han and L. Niu, *Adv. Mater.*, 2015, **27**, 6906–6913.
- 185 T. Di, B. Zhu, B. Cheng, J. Yu and J. Xu, *J. Catal.*, 2017, **352**, 532–541.
- 186 Y. Liu, G. Ji, M. A. Dastageer, L. Zhu, J. Wang, B. Zhang, X. Chang and M. A. Gondal, *RSC Adv.*, 2014, **4**, 56961–56969.
- 187 G. Song, F. Xin and X. Yin, *J. Colloid Interface Sci.*, 2015, **442**, 60–66.
- 188 L. Kuai, Y. Zhou, W. Tu, P. Li, H. Li, Q. Xu, L. Tang, X. Wang, M. Xiao and Z. Zou, *RSC Adv.*, 2015, **5**, 88409–88413.
- 189 J. Q. Jiao, Y. C. Wei, Z. Zhao, W. J. Zhong, J. Liu, J. M. Li, A. J. Duan and G. Y. Jiang, *Catal. Today*, 2015, **258**, 319–326.
- 190 A. Ayati, A. Ahmadpour, F. F. Bamoharram, M. M. Heravi and M. Sillanpää, *Gold Bull.*, 2012, **45**, 145–151.
- 191 M. Gui, W. M. P. Wong, S. Chai and A. R. Mohamed, *Chem. Eng. J.*, 2015, **278**, 272–278.
- 192 C. Wang, K. M. Kwon, M. L. Odlyzko, B. H. Lee and M. Shim, *J. Phys. Chem. C*, 2007, **111**, 11734–11741.
- 193 C. Wang, R. L. Thompson, J. Baltrus and C. Matranga, *J. Phys. Chem. Lett.*, 2010, **1**, 48–53.
- 194 L. C. Sim, K. H. Leong, P. Saravanan and S. Ibrahim, *Appl. Surf. Sci.*, 2015, **358**, 122–129.
- 195 A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183–191.
- 196 W. Ong, L. Tan, S. Chai and S. Yong, *Chem. Commun.*, 2015, **51**, 858–861.
- 197 Y. Zhang, Z. Tang, X. Fu and Y. Xu, *ACS Nano*, 2010, **4**, 7303–7314.
- 198 W. Fan, Q. Zhang and Y. Wang, *Phys. Chem. Chem. Phys.*, 2013, **15**, 2632–2649.
- 199 A. A. Balandin, S. Ghosh, W. Bao, I. Calizo, D. Teweldebrhan, F. Miao and C. N. Lau, *Nano Lett.*, 2008, **8**, 902–907.
- 200 L. Tan, S. Chai and A. R. Mohamed, *ChemSusChem*, 2012, **5**, 1868–1882.
- 201 M. D. Stoller, S. J. Park, Y. Zhu, J. An and R. S. Ruoff, *Nano Lett.*, 2008, **8**, 3498–3502.
- 202 H. K. Chae, D. Y. Siberio-perez, J. Kim, Y. B. Go, M. Eddaoudi, A. J. Matzger, M. O’Keeffe and O. M. Yaghi, *Nature*, 2004, **427**, 523–527.
- 203 X. Li, J. Yu, S. Wageh, A. A. Al-Ghamdi and J. Xie, *Small*, 2016, **12**, 6640–6696.
- 204 L. S. Schadler, S. C. Giannaris and P. M. Ajayan, *Appl. Phys. Lett.*, 1998, **73**, 3842–3844.
- 205 X. Du, I. Skachko, A. Barker and E. Y. Andrei, *Nat. Nanotechnol.*, 2008, **3**, 491–495.
- 206 K. I. Bolotin, K. J. Sikes, Z. Jiang, M. Klima, G. Fudenberg, J. Hone, P. Kim and H. L. Stormer, *Solid State Commun.*, 2008, **146**, 351–355.
- 207 R. R. Nair, P. Blake, A. N. Grigorenko, K. S. Novoselov, T. J. Booth, T. Stauber, N. M. R. Peres and A. K. Geim, *Science*, 2008, **320**, 1308.
- 208 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666–669.

- 209 F. Li, L. Zhang, J. Tong, Y. Liu, S. Xu, Y. Cao and S. Cao, *Nano Energy*, 2016, **27**, 320–329.
- 210 W. Ong, L. Tan, S. Chai, S. Yong and A. R. Mohamed, *Nano Res.*, 2014, **7**, 1528–1547.
- 211 K. P. Loh, Q. Bao, G. Eda and M. Chhowalla, *Nat. Chem.*, 2010, **2**, 1015–1024.
- 212 Y. Zhang, N. Zhang, Z. Tang and Y. Xu, *J. Phys. Chem. C*, 2014, **118**, 5299–5308.
- 213 M. Yang, X. Pan, N. Zhang and Y. Xu, *CrystEngComm*, 2013, **15**, 6819–6828.
- 214 Y. Sun, Q. Wu and G. Shi, *Energy Environ. Sci.*, 2011, **4**, 1113–1132.
- 215 L. Han, P. Wang and S. Dong, *Nanoscale*, 2012, **4**, 5814–5825.
- 216 N. Zhang, Y. Zhang and Y. Xu, *Nanoscale*, 2012, **4**, 5792–5813.
- 217 G. Williams, B. Seger and P. V. Kamat, *ACS Nano*, 2008, **2**, 1487–1491.
- 218 L. Tan, W. Ong, S. Chai and A. R. Mohamed, *Nanoscale Res. Lett.*, 2013, **8**, 465.
- 219 L. Tan, W. Ong, S. Chai, B. T. Goh and A. R. Mohamed, *Appl. Catal., B*, 2015, **179**, 160–170.
- 220 L. Tan, W. Ong, S. Chai and A. R. Mohamed, *Appl. Catal., B*, 2015, **166–167**, 251–259.
- 221 Y. Zhang, D. Ma, Y. Zhang, W. Chen and S. Huang, *Nano Energy*, 2013, **2**, 545–552.
- 222 D. Mosconi, D. Mazzier, S. Silvestrini, A. Privitera, C. Marega, L. Franco and A. Moretto, *ACS Nano*, 2015, **9**, 4156–4164.
- 223 M. Li, M. Wang, L. Zhu, Y. Li, Z. Yan, Z. Shen and X. Cao, *Appl. Catal., B*, 2018, **231**, 269–276.
- 224 P. Li, S. Ouyang, G. Xi, T. Kako and J. Ye, *J. Phys. Chem. C*, 2012, **116**, 7621–7628.
- 225 P. Li, S. Ouyang, Y. Zhang, T. Kako and J. Ye, *J. Mater. Chem. A*, 2013, **1**, 1185–1191.
- 226 P. Li, H. Xu, L. Liu, T. Kako, N. Umezawa, H. Abe and J. Ye, *J. Mater. Chem. A*, 2014, **2**, 5606–5609.
- 227 D. Dambournet, I. Belharouak and K. Amine, *Chem. Mater.*, 2010, **22**, 1173–1179.
- 228 P. Wang, Y. Bai, J. Liu, Z. Fan and Y. Hu, *Catal. Commun.*, 2012, **29**, 185–188.
- 229 G. H. Li, S. Ciston, Z. V. Saponjic, L. Chen, N. M. Dimitrijevic, T. Rajh and K. A. Gray, *J. Catal.*, 2008, **253**, 105–110.
- 230 L. Chen, M. E. Graham, G. Li, D. R. Gentner, N. M. Dimitrijevic and K. A. Gray, *Thin Solid Films*, 2009, **517**, 5641–5645.
- 231 J. Jiao, Y. Wei, Y. Zhao, Z. Zhao, A. Duan, J. Liu, Y. Pang, J. Li, G. Jiang and Y. Wang, *Appl. Catal., B*, 2017, **209**, 228–239.
- 232 Q. D. Truong, T. Le, J. Liu, C. Chung and Y. Ling, *Appl. Catal., A*, 2012, **437–438**, 28–35.
- 233 K. Tomita, V. Petrykin, M. Kobayashi, M. Shiro, M. Yoshimura and M. Kakihana, *Angew. Chem., Int. Ed.*, 2006, **45**, 2378–2381.
- 234 H. Zhao, L. Liu, J. M. Andino and Y. Li, *J. Mater. Chem. A*, 2013, **1**, 8209–8216.
- 235 J. Yu, J. Low, W. Xiao, P. Zhou and M. Jaroniec, *J. Am. Chem. Soc.*, 2014, **136**, 8839–8842.
- 236 M. S. Akple, J. Low, Z. Qin, S. Wageh, A. A. Al-Ghamdi, J. Yu and S. Liu, *Chin. J. Catal.*, 2015, **36**, 2127–2134.
- 237 Z. Xiong, Y. Luo, Y. Zhao, J. Zhang, C. Zheng and J. C. S. Wu, *Phys. Chem. Chem. Phys.*, 2016, **18**, 13186–13195.
- 238 Y. Zhao, Y. Wei, X. Wu, H. Zheng, Z. Zhao, J. Liu and J. Li, *Appl. Catal., B*, 2018, **226**, 360–372.
- 239 Q. D. Truong, H. T. Hoa and T. S. Le, *J. Colloid Interface Sci.*, 2017, **504**, 223–229.
- 240 J. Y. Do, V. Tamilavan, R. Agneeswari, M. H. Hyun and M. Kang, *J. Photochem. Photobiol., A*, 2016, **330**, 30–36.
- 241 A. Crake, K. C. Christoforidis, A. Kafizas, S. Zafeiratos and C. Petit, *Appl. Catal., B*, 2017, **210**, 131–140.
- 242 S. Kumar, M. A. Isaacs, R. Trofimovaite, L. Durndell, C. M. A. Parlett, R. E. Douthwaite, B. Coulson, M. C. R. Cockett, K. Wilson and A. F. Lee, *Appl. Catal., B*, 2017, **209**, 394–404.
- 243 Y. Izumi, *Coord. Chem. Rev.*, 2013, **257**, 171–186.
- 244 K. Mori, H. Yamashita and M. Anpo, *RSC Adv.*, 2012, **2**, 3165–3172.
- 245 K. Li, X. An, K. H. Park, M. Khraisheh and J. Tang, *Catal. Today*, 2014, **224**, 3–12.
- 246 P. Pan and Y. Chen, *Catal. Commun.*, 2007, **8**, 1546–1549.
- 247 M. S. Hamdy, R. Amrollahi, I. Sinev, B. Mei and G. Mul, *J. Am. Chem. Soc.*, 2014, **136**, 594–597.
- 248 W. Tu, Y. Zhou and Z. Zou, *Adv. Mater.*, 2014, **26**, 4607–4626.
- 249 X. Chang, T. Wang and J. Gong, *Energy Environ. Sci.*, 2016, **9**, 2177–2196.